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TECHNICAL NOTE

No. 1760

NACA AND OFFICE OF NAVAL RESEARCH METALLURGICAL

INVESTIGATION OF TWO LARGE FORGED DISCS

OF S-590 ALLOY

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Washington

February 1949

1949
TECHNICAL NOTE
1760

8208

3.17.90/91



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OF TWO LARGE FORGED DISCS OF S-590 ALLOY

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SUMMARY

The properties of large forged discs of S-590 alloy at room temperature, 1200°, 1350°, and 1500° F were studied in order to determine the level of properties obtainable in forgings of the type required for the rotor discs of gas turbines. One disc was aged after forging. The other was solution-treated and aged. A limited amount of testing was carried out on the solution-treated disc prior to aging. The data reported include the results of tensile, impact, rupture, time-deformation, creep, and structural-stability tests.

The high physical properties of the forged and aged disc at temperatures up to 1350° F were its outstanding characteristic. The solution-treated and aged disc had by far the best properties at 1500° F and, except at short time periods, was considerably better at 1350° F. Somewhat higher rupture, total-deformation, and creep strengths for time periods up to at least 2000 hours were obtained at 1200° F from the forged and aged disc. No great difference, on the basis of limited tests at 1200° and 1350° F, resulted from testing the solution-treated disc before aging.

The properties of specimens cut from different locations in the discs varied somewhat. However, the uniformity was good for the type of forging made from such a highly alloyed material.

Both the forged and aged and solution-treated and aged materials were structurally unstable during creep and rupture testing. The latter treatment, however, resulted in the best retention of properties over long periods of time at high temperatures.

The properties of the solution-treated and aged disc were similar to those reported for bar stock with the same heat treatments. This indicates that the properties should be reasonably reproducible in discs up to the size considered in this investigation. Reproduction of the properties of the as-forged and aged disc would probably require a considerable degree of control of forging practice.

The work on which this report is based is part of a cooperative investigation of several heat-resistant alloys in the form of large forged discs. The properties of the S-590 discs are compared in this report with those obtained for similar discs of S-816 alloy.

INTRODUCTION

This report presents the results of a study of the room-temperature, 1200°, 1350°, and 1500° F properties of two large discs of S-590 alloy. One of the discs was tested in the as-forged and aged condition. The other disc was tested to a limited extent after only a solution treatment; and more completely tested as solution-treated and aged.

The primary purpose of this study was to determine the level of properties exhibited by S-590 alloy in the form of large forgings of the type required for rotor discs of gas turbines and to determine the relative properties of such discs as-forged and aged and as-solution-treated and aged. The S-590 alloy discs, for which properties are given in this report, were two of a series of similar discs of several alloys now being studied. The results obtained previously from similar investigations on 19-9DL, CSA, low-carbon N-155, Timken, and EME discs are contained in references 1 to 9.

The work on the disc materials is being carried out as part of two correlated programs of research on alloys for gas-turbine applications in progress in this country. The National Advisory Committee for Aeronautics is sponsoring work directed toward the development of improved high-temperature alloys for gas turbines used in aircraft power plants. A concurrent program, formerly sponsored by the National Defense Research Committee, Office of Scientific Research and Development, and now sponsored by the Office of Naval Research, Navy Department, is being directed to the development of alloys for gas-turbine applications in general and, in particular, to both ship and aircraft propulsion. The work herein was performed with the financial assistance of the National Advisory Committee for Aeronautics and the Office of Naval Research, Navy Department.

This report is based on the joint effort of the cooperating research programs and is being distributed by both the NACA and the Navy. The investigation of these discs for the NACA was conducted at the Engineering Research Institute of the University of Michigan and for the Navy by Battelle Memorial Institute.

TEST MATERIALS

The code number assigned to the discs was NR-74B. The as-forged and aged disc was designated NR-74B-F; the solution-treated disc, NR-74B-Q; and the aged portion of the solution-treated disc, NR-74B-QA.

The available information describing the two discs may be summarized as follows:

Manufacturer:

Allegheny-Ludlum Steel Corporation

Heat number:

41582

Chemical composition:

| <u>C</u> | <u>Mn</u> | <u>Si</u> | <u>P</u> | <u>S</u> | <u>Cr</u> | <u>Ni</u> | <u>Co</u> | <u>Mo</u> | <u>W</u> | <u>Cb</u> | <u>Fe</u> |
|----------|-----------|-----------|----------|----------|-----------|-----------|-----------|-----------|----------|-----------|-----------|
| 0.45 | 1.44 | 0.56 | 0.015 | 0.018 | 19.76 | 19.05 | 20.20 | 4.03 | 4.08 | 3.35 | Remainder |

Fabrication procedure:

A 12-inch ingot was poured from a 2-ton electric-arc furnace. The 12-inch ingot was hammer clogged from 2250° F to a 9-inch-square billet which was air-cooled and ground. Two portions of this billet were then upset forged from 2250° F to rough 4-inch-thick discs. All hot-working was with a flat die on a 12,000-pound hammer. The finishing and heat treatments for the individual discs were as follows:

| <u>Disc designation</u> | <u>Finish forging procedure</u> | <u>Heat treatment</u> |
|-------------------------|---|--|
| NR-74B-F | The disc was reforged from 2250° F to $3\frac{3}{4}$ inches thick and cooled. Then it was reforged from 2000° F in one heat to $3\frac{3}{8}$ inches thick (10-percent reduction) by 18 inches in diameter. | Aged for 16 hours at 1400° F and air-cooled. |
| NR-74B-Q | The disc was reforged from 2250° F to 3 inches thick by 18 inches in diameter. | Solution-treated for $3\frac{1}{4}$ hours at 2300° F and water-quenched. |
| NR-74B-QA | Same forging as NR-74B-Q. Coupons of disc NR-74B-Q which were aged prior to testing were designated NR-74B-QA. | Coupons from NR-74B-Q were aged for 16 hours at 1400° F and air-cooled. |

Sampling:

One-half of each of the two discs, NR-74B-F and NR-74B-Q, was supplied for the present study, one-quarter of each disc going to the University of Michigan and Battelle Memorial Institute, respectively. Figures 1 and 2 show the location of the samples cut from the halves of both discs and the code system identifying the coupons. The numerals refer to locations on the flat faces of the discs, and the letters refer to the locations through the thickness of the discs.

EXPERIMENTAL PROCEDURE

The investigation was designed to provide the following information:

- (1) The physical properties at room temperature, 1200°, 1350°, and 1500° F which can be expected in large forgings of the S-590 alloy analysis;
- (2) the effect of fabrication and heat treatment on these physical properties;
- (3) the variation in properties which might be present in various locations in such large forgings; and
- (4) the change in room-temperature properties resulting from exposure to elevated temperatures under stress for prolonged time periods.

The physical-property data obtained for these large forged discs of S-590 alloy included short-time tensile properties, impact strengths, rupture test characteristics, design curves of stress against time for total deformations of 0.1, 0.2, 0.5, and 1.0 percent at 1200°, 1350°, and 1500° F, and creep characteristics. The curves of stress against time for total deformation were obtained from curves of elongation against time from both stress-rupture and creep tests.

The uniformity of the disc materials was checked by means of a hardness survey and to a limited extent by tensile and rupture tests on coupons from representative locations throughout the discs. Hardness, tensile, and impact tests and metallographic examinations on specimens after completion of the tests were used to estimate the stability of the material during prolonged exposure to temperature and stress.

The testing procedures used for the short-time tension, stress-rupture, and creep tests were in accordance with the provisions of the A.S.T.M. Recommended Practices E21-43 and E22-41.

RESULTS

The data obtained from the S-590 discs are presented as a series of tables and figures which show the hardness, impact, tensile, rupture, time-deformation, creep, and stability characteristics. The principal results on the discs with three types of treatment are summarized in figures 3 and 4.

Hardness Survey

Results of hardness tests on the original materials are given in table I and figure 5. The surveys indicated that the hardness increase was only slight from the center to the rim of the discs. The material at the flat surfaces of the discs was considerably harder than the material in the interior of the discs.

The as-forged and aged disc, NR-74B-F, had a hardness range of 235 to 311 as compared with a range of 190 to 235 Brinell hardness for the solution-treated disc, NR-74B-Q. Although no hardness survey was made on the solution-treated and aged disc material, NR-74B-QA, hardness tests indicated that aging the solution-treated material increased the hardness by approximately 30 Brinell points.

Short-Time Tensile Properties

The results of the short-time tensile tests at room temperature, 1200°, 1350°, and 1500° F are summarized in table I.

The tensile strengths of the as-forged and aged disc, NR-74B-F, were in general somewhat higher, while its yield strengths were markedly higher at both room temperature and 1200° F than those of the heat-treated disc. At 1350° F the forged and aged disc had similar tensile strength but higher yield strength than the solution-treated and aged disc. On the basis of only one test the as-solution-treated material had higher tensile strength but similar yield strength to the material aged after solution treatment. At 1500° F the solution-treated and aged material, NR-74B-QA, was slightly stronger than the as-forged and aged material. A brief résumé of comparative tensile properties taken from table I is given in the following tabulation:

| Disc | Temperature (°F) | Tensile strength (psi) | 0.2-percent-offset yield strength (psi) | Elongation (percent) |
|-----------|---------------------|------------------------------|---|-------------------------|
| NR-74B-F | 75 | 129,050 | 98,250 | 8 |
| NR-74B-Q | 75 | 119,500 | 57,000 | 36 |
| NR-74B-QA | 75 | 130,500 | 70,500 | 17 |
| NR-74B-F | 1200 | 88,700 | 71,750 | 15 |
| NR-74B-Q | 1200 | 82,000 | 44,000 | 12 |
| NR-74B-QA | 1200 | 81,600 | 49,000 | 27 |
| NR-74B-F | 1350 | 64,625 | 55,000 | 29 |
| NR-74B-Q | 1350 | 71,250 | 46,000 | 11 |
| NR-74B-QA | 1350 | 65,750 | 46,000 | 25 |
| NR-74B-F | 1500 | 43,125 | 35,900 | 25 |
| NR-74B-QA | 1500 | 44,400 | 37,850 | 18 |

At room temperature the solution-treated disc, NR-74B-Q, had the highest ductility and the as-forged and aged disc, NR-74B-F, the lowest ductility. The reverse ductility comparison was true at high temperature. Aging the solution-treated disc for 16 hours at 1400° F caused a substantial decrease in room-temperature tensile ductility but resulted in just as marked an improvement in ductility at temperatures of 1200° F and above.

The properties of specimens from various locations in the discs were quite uniform. Specimens taken tangentially from the as-forged and aged disc had higher strengths than the radial specimens. No such strength difference was observed between radial and tangential specimens of the solution-treated disc. Because of lack of material no consistent comparison was possible for material taken radially near the surface and center material. However, what data there were indicated good uniformity.

Charpy Impact Resistance

Charpy impact resistance (V-notch) was determined on specimens from two discs, NR-74B-F and NR-74B-QA. Data are shown in table II and figures 3 and 4 for tests at room temperature, 1200°, 1350°, and 1500° F after holding at temperature for a time period sufficiently long to insure a uniform temperature in the specimen.

The Charpy impact resistance of the solution-treated and aged disc was slightly higher at all test temperatures than that of the forged and aged disc. For both discs, there was a slight increase in impact resistance with temperature. Specimens from near the flat surfaces of both discs had higher impact resistance than interior specimens at all temperatures.

Rupture Test Characteristics

The stress-rupture data for the tests at 1200°, 1350°, and 1500° F are shown in table III, and the rupture strengths obtained from the curves of stress against rupture time in figure 6 are summarized in table IV. Rupture ductilities at various time periods are also given in table IV. All specimens tested except one were radial specimens, located as indicated in table III.

There was very little difference in rupture strengths between the three conditions of treatment for the discs at 1200° F. The solution-treated and aged disc, NR-74B-QA, did show a slight superiority at time periods of 1000 hours and longer, its 100- and 1000-hour rupture strengths being 52,000 and 42,000 psi, respectively.

At 1350° F the solution-treated discs were definitely superior to the forged and aged disc at 100 hours and longer. Comparative rupture strengths were 32,000 psi for NR-74B-QA and 27,500 psi for NR-74B-F at 100 hours and 25,000 psi for NR-74B-QA compared with 18,000 psi for NR-74B-F at 1000 hours. Aging the solution-treated disc for 16 hours at 1400° F did not affect rupture strengths at 1350° F.

The solution-treated and aged disc was much stronger than the forged and aged disc at 1500° F. The comparative 100-hour rupture strengths for the two discs were 20,000 and 13,100 psi and 1000-hour strengths were 15,000 and 6,000 psi.

Inspection of the curves of stress against rupture time in figure 6 indicates little change in the slope of the curves with increased temperature of testing for the solution-treated disc. The increased slope of the curves for the forged and aged disc with increasing temperature accounts for its lower strength. This difference clearly indicates the beneficial effect of a solution treatment on properties at temperatures above 1200° F.

The rupture tests on specimens from various locations in the discs indicated that the disc material was fairly uniform and that, if anything, the material taken radially near the rim in the center plane tended to be weaker than material from other locations. Thus, since most of the material tested came from this location, the results obtained were probably conservative for the properties of the disc as a whole.

Rupture test ductilities shown were better for the solution-treated and aged disc than for the forged and aged disc in all cases. Aging the solution-treated material produced a marked improvement in rupture ductility at 1200° F but had little effect on ductility at 1350° F. Actually, the ductility of all the material was good, being at least 5 percent for fracture in 1000 hours.

Time-Deformation Characteristics

A convenient method of describing the high-temperature strength of a material is curves of stress against the time required for various total deformations. Deformation data from both stress-rupture and creep tests are used to prepare such design curves. This information, along with the curves of stress against rupture time, gives a fairly complete picture of the expected performance of an alloy under conditions of constant tensile stress. The time-deformation data obtained on the S-590 discs in three conditions are plotted on semilogarithmic coordinates in figures 7 to 14 for total deformations of 0.1, 0.2, 0.5, and 1.0 percent at 1200°, 1350°, and 1500° F for time periods up to 2000 hours. Additional curves which indicate the time of transition from a minimum creep rate to the increasing rate of third-stage creep have been added so as to show where rapid elongation preceding failure starts.

The curves of stress against time for total deformation were plotted from the data in tables V, VI, and VII. These data were taken from the curves of elongation against time for the rupture and creep tests. Somewhat erratic data resulted from the tests. Sufficient check tests were made, however, to demonstrate that these erratic results were due to a variation between specimens from different locations in the discs. The actual curves of elongation against time have not been included in this report.

The stresses to cause various total deformation in 1, 10, 100, 1000, and 2000 hours, as defined by the curves in figures 7 to 14, are given in tables VIII, IX, and X. The most pronounced difference between discs was found at 1500° F where the solution-treated and aged disc had deformation strengths from two to three times higher than those of the forged and aged disc. The difference between the discs was much less at 1200° and 1350° F. The forged and aged disc gave strengths somewhat higher than the solution-treated disc and both were higher than the solution-treated and aged material at 1200° F, particularly at 0.5-percent total deformation. At 1350° F the solution-treated and aged material had higher strengths, the degree of superiority increasing with the amount of total deformation considered.

Creep Strengths

Many engineers are accustomed to base designs on creep rates, especially for long periods of service. For this reason, the creep rate data have been collected from the curves of elongation against time and are shown in table XI for creep tests and table III for rupture tests. The logarithmic curves of stress against creep rate for the tests at 1200°, 1350°, and 1500° F on the S-590 discs are shown in figure 15.

The creep rates plotted were either minimum rates or final rates from 1000-hour tests at 1200° F and 2000-hour tests at 1350° and 1500° F. The creep strengths obtained from figure 15 were as follows:

| Disc | Temperature (°F) | Stress (psi) for creep rates of - | |
|-----------|---------------------|-----------------------------------|--------------------|
| | | 0.0001 percent/hr | 0.00001 percent/hr |
| NR-74B-F | 1200 | 27,500 | ----- |
| NR-74B-QA | 1200 | 23,000 | ----- |
| NR-74B-F | 1350 | 10,600 | ----- |
| NR-74B-QA | 1350 | 16,400 | 12,100 |
| NR-74B-F | 1500 | 8,800 | ----- |
| NR-74B-QA | 1500 | 10,000 | 7,100 |

^aEstimated strength.

It is observed that at temperatures above 1200° F the solution-treated and aged disc, NR-74B-QA, is much superior to the forged and aged disc, NR-74B-F.

These creep strengths can be compared with the deformation strengths in tables VIII, IX, and X. The creep strengths for a rate of 0.0001 percent per hour at 1200° F are apparently safe for use for time periods up to 10,000 hours since extrapolation of the curves of transition to third-stage creep in figures 7 and 9 to 10,000 hours indicates that at the creep strengths listed second-stage creep will still prevail.

At 1350° and 1500° F extrapolation of the transition curves of figures 12 and 14 to 10,000 hours gives stresses about the same as those producing a creep rate of 0.0001 percent per hour for the solution-treated and aged disc, NR-74B-QA. This is not true for the forged and aged disc, NR-74B-F, transition to third-stage creep occurring in approximately 2000 hours under stresses causing a creep rate of 0.0001 percent per hour. (See figs. 10 and 13.) This means that the reported creep strength for NR-74B-F at these higher temperatures would not be suitable as a basis for design for longer time periods than 2000 hours, while the creep strengths of NR-74B-QA can be used, with caution, out to 10,000 hours.

At 1200° F the data were not sufficient to define the strengths for a creep rate of 0.00001 percent per hour. At higher temperatures the slopes of the curves of stress against rupture time indicate that creep strengths for this rate would not be suitable as a basis for design for prolonged time periods for the forged and aged disc and that caution should be observed when extended service periods are contemplated for solution-treated and aged material.

Stability Characteristics

Some of the completed-test specimens from each of the discs were subjected to tensile, impact, and hardness tests at room temperature, after creep testing at 1200°, 1350°, and 1500° F, with the results shown in table XII.

The most significant property changes observed as a result of creep testing were the decreases in impact resistance and tensile test ductility at room temperature. Impact strengths were low initially and were very low after creep testing. The decrease in ductility was even more pronounced than that of impact strength.

There was no significant change in hardness as a result of creep testing for the forged and aged disc, but the solution-treated and aged disc increased in hardness during testing. The tensile-test strength properties of the forged and aged disc, NR-74B-F, decreased progressively with increasing creep test temperature. Those of the as-solution-treated material, NR-74B-Q, increased as a result of a 1200° F creep test, while the strengths of the solution-treated and aged disc, NR-74B-QA, were higher after creep tests at 1200° and 1350° F, but were lower in strength than the original material after a 1500° F creep test.

Photomicrographs of the structures of the original materials and after creep and rupture testing are shown in figures 16 to 22. The forged and aged disc, NR-74B-F, had nonuniform structure as evidenced by grain-size differences and distribution of the excess constituents. (See fig. 16.) These differences were also observed in the structure of some of the completed-test specimens.

Only a small amount of general precipitation was observed in the forged and aged disc as a result of creep and rupture testing at 1200° F. (See figs. 17(a) and 18(a).) Considerable agglomeration occurred during testing at 1350° F. The differences in amount of precipitate between the creep and rupture test specimens, shown by comparison of figures 17(b) and 18(b), were another indication of nonuniformity of material. Further agglomeration of the precipitated phases was observed in the 1500° F rupture specimen. (See fig. 18(c).)

The original microstructures of the solution-treated disc, NR-74B-Q, and the solution-treated and aged disc, NR-74B-QA, (fig. 19) were different in that more precipitates were present in the aged material. Heavy general precipitation occurred during rupture and creep testing of both materials and agglomeration increased as the test temperature was increased. The precipitation did not appear quite so heavy in the creep specimens as in the rupture specimens.

Fracture of the longest-time rupture specimens of the forged and aged disc appear to be both transgranular and intergranular while those of the solution-treated discs were largely intergranular.

Allegheny-Ludlum Data on NR-74B Discs

Table XIII gives the available results from tensile, hardness, and rupture tests obtained by the Allegheny-Ludlum Steel Corporation on the other halves of these S-590 discs. Also listed are comparative results obtained in this investigation. In general, the comparative results show good agreement.

CONCLUDING REMARKS

In general, the solution-treated and aged disc had the best properties at high temperatures. At 1200° F the forged and aged disc had better rupture strengths out to 100 hours and higher total-deformation strengths to at least 2000 hours. At 1350° and 1500° F the solution-treated and aged disc was definitely superior in properties. On the basis of a limited number of tests at 1200° and 1350° F, the properties of a plain solution-treated disc were almost the same as those for the solution-treated and aged disc material.

The as-forged and aged disc had much higher yield strength at room temperature, 1200°, and 1350° F than the solution-treated and aged disc. This characteristic might be important in applications involving high stresses at low temperatures at the centers of rotor discs or in applications involving high stresses for short time periods up to 1350° F.

The data reported by the Allegheny-Ludlum Steel Corporation show that aging the as-forged disc at 1400° F reduced properties at room temperature and probably increased rupture strength at 1350° F. Increasing the aging temperature to 1500° F further reduced yield strength at room temperature and lowered rupture strength below that of the material aged at 1400° F. Their data also show that aging at 1500° F after a solution treatment results in somewhat lower rupture strength at 1500° F than aging at 1400° F, at least for time periods longer than about 100 hours.

Table XIV has been prepared to show the comparative properties of solution-treated and aged bar stock of S-590 alloy and large discs. The tensile properties of bar stock were somewhat higher than a similarly treated disc. Rupture properties at 1350° and 1500° F and total-deformation properties at 1500° F agree quite well for the solution-treated and aged bar stock and the disc, an indication of the possibility of good reproducibility of high-temperature properties in different forms for S-590 alloy.

The properties of the discs had, in general, good uniformity for such large forgings of highly alloyed material. Wide variations in grain size and microstructure did not appear to affect properties greatly, except to cause erratic data for the studies of stress against time for total deformation. Such variations as were present were reduced somewhat by the solution and aging treatment.

A major problem in using data of the type obtained in this investigation is to estimate the degree of reproducibility. Experience with other high-alloy steels indicates that fairly good control over forging practice would be required to reproduce consistently the properties of the as-forged and aged disc. The agreement in properties between bar stock and the disc when solution-treated and aged suggests that the properties of discs should be fairly reproducible when heat-treated. Until more data on the properties of discs made by this and, especially, other fabrication procedures are available, it should be assumed that the data herein reported apply only to the particular discs tested and fabricated and heat-treated in the manner indicated.

The heat treatments used on the discs covered by this investigation were based on a large amount of experimental work by the Allegheny-Ludlum Steel Corporation. Deviation from these conditions would result in pronounced changes in properties at high temperatures.

Table XV presents a summary of the comparative properties of discs of two alloys, S-590 and S-816 (see reference 10), studied at 1200°, 1350°, and 1500° F in the cooperative research program. This comparison shows, in general, that for similarly treated material the S-816 alloy disc has better properties than the S-590 alloy disc.

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TABLE I

SHORT-TIME TENSILE PROPERTIES OF B-590 ALLOY DISCS NR-74B

[NACA data except where indicated. All tensile tests were made on standard 0.505-in.-diameter specimens]

| Disc (a) | Specimen number | Specimen location (b) | Temperature (°F) | Tensile strength (psi) | Offset yield strengths (psi) | | | Proportional limit (psi) | Elongation in 2 in. (percent) | Reduction of area (percent) | Brinell hardness | Modulus of elasticity | |
|-----------|-----------------|-----------------------|------------------|------------------------|------------------------------|-------------|-------------|--------------------------|-------------------------------|-----------------------------|------------------|------------------------|------|
| | | | | | 0.02 percent | 0.1 percent | 0.2 percent | | | | | | |
| NR-74B-F | 14Y | CRR | 75 | 130,600 | 69,000 | 91,000 | 100,000 | 37,500 | 9 | 11.5 | 267 | 30.5 × 10 ⁶ | |
| | 14X | SRR | 75 | 127,900 | 72,500 | 89,000 | 96,500 | 47,500 | 7 | 9.2 | 267-302 | 29.9 | |
| | 16Y | CTR | 75 | 137,750 | 75,000 | 99,000 | 108,000 | 37,500 | 11.5 | 23.2 | 293 | 28.8 | |
| | 16X | STR | 75 | 147,000 | 91,000 | 110,000 | 120,000 | 50,000 | 10 | 20.6 | 311 | 30.3 | |
| | 12Z | SHR | 1200 | 87,900 | ----- | 62,500 | 70,000 | 22,500 | 16.5 | 21.3 | ----- | 23.8 | |
| | 14Z | SRR | 1200 | 89,500 | ----- | 70,000 | 73,500 | 40,000 | 14 | 23.0 | ----- | 22.1 | |
| | 13Z | SHR | 1350 | 65,375 | ----- | 48,500 | 53,000 | 17,500 | 31 | 45.3 | ----- | 20.3 | |
| | 15X | SRC | 1350 | 63,875 | ----- | 53,000 | 57,000 | 22,500 | 27 | 35.0 | ----- | 19.6 | |
| | 02Y | CRR | 1500 | 43,250 | ----- | 32,500 | 36,000 | ----- | 20 | 29.8 | ----- | 14.0 | |
| | 01Z | SRR | 1500 | 43,000 | ----- | 30,200 | 35,750 | ----- | 31 | 38.2 | ----- | 19.4 | |
| | NR-74B-Q | 15Y | CRR | 75 | 117,500 | 37,500 | 53,000 | 59,000 | 22,500 | 30.5 | 27.2 | 211 | 27.6 |
| | | 16Z | SRC | 75 | 121,500 | 35,000 | 48,500 | 55,000 | 22,500 | 42 | 41.3 | 215 | 23.5 |
| 17Y | | CTR | 75 | 119,000 | 40,000 | 53,500 | 59,000 | 25,000 | 38 | 37.2 | 223 | 29.4 | |
| 17X | | STR | 75 | 121,000 | 30,000 | 47,000 | 57,000 | 17,500 | 40 | 36.5 | 218 | 27.6 | |
| 14Z | | SRR | 1200 | 82,000 | ----- | 41,500 | 44,000 | 20,000 | 12 | 17.7 | ----- | 22.2 | |
| 15Z | | SHR | 1350 | 71,250 | ----- | 41,500 | 46,000 | 22,500 | 11 | 13.0 | ----- | 22.2 | |
| NR-74B-QA | 16Y | CRC | 75 | 130,500 | 45,000 | 63,500 | 70,500 | 25,000 | 17 | 18.2 | 259 | 27.8 | |
| | 16X | SRC | 1200 | 81,600 | ----- | 46,000 | 49,000 | 27,500 | 27 | 31.2 | ----- | 23.8 | |
| | 13X | SRR | 1350 | 65,750 | ----- | 43,500 | 46,000 | 20,000 | 25 | 30.4 | ----- | 22.8 | |
| | 01Y | CRR | 1500 | 44,500 | ----- | 35,700 | 38,250 | ----- | 22.7 | 27.2 | ----- | 18.0 | |
| | 02Z | SRR | 1500 | 44,250 | ----- | 34,400 | 37,450 | ----- | 13.3 | 15.9 | ----- | 19.5 | |

^aHeat treatments:

NR-74B-F: As-forged; 16 hr at 1400° F.

NR-74B-Q: 2300° F water-quenched.

NR-74B-QA: 2300° F water-quenched; 16 hr at 1400° F.

^bCRR center-plane radial specimen near rim of disc.

SRR surface-plane radial specimen near rim of disc.

CTR center-plane tangential specimen near rim of disc.

STR surface-plane tangential specimen near rim of disc.

SRC surface-plane radial specimen near center of disc.

CRC center-plane radial specimen near center of disc.

^cNDRC and Navy data at 1500° F.

NACA

TABLE II

CHARPY NOTCH-BAR IMPACT RESISTANCE AT ROOM TEMPERATURE, 1200°,
1350°, and 1500° F FOR S-590 ALLOY DISCS NR-74B

[NDRC and Navy data]

| Disc (1) | Specimen number | Specimen location | Test temperature (°F) | Charpy impact strength (ft-lb) |
|-------------|--------------------|----------------------|-----------------------------|--------------------------------------|
| NR-74B-F | 5C | Interior | Room | 5 |
| | 7C | Interior | | 5 |
| | 8B | Interior | | 5 |
| | 5A | Surface | | 6 |
| | 7F | Surface | | 7 |
| NR-74B-F | 5D | Interior | 1200 | 8 |
| | 7D | Interior | | 8 |
| | 8E | Interior | | 8 |
| | 5F | Surface | | 13 |
| | 10A | Surface | | 10 |
| NR-74B-F | 8D | Interior | 1350 | 8 |
| | 6D | Interior | | 10 |
| | 10E | Interior | | 8 |
| | 6A | Surface | | 14 |
| | 8A | Surface | | 13 |
| | 8F | Surface | | 13 |
| NR-74B-F | 6C | Interior | 1500 | 12 |
| | 8C | Interior | | 10 |
| | 5B | Interior | | 13 |
| | 6F | Surface | | 18 |
| | 7A | Surface | | 17 |
| | 10A | Surface | | 14 |
| NR-74B-QA | 8D | Interior | Room | 6 |
| | 7E | Interior | | 10 |
| | 8F | Surface | | 10 |
| NR-74B-QA | 9C | Interior | 1200 | 10 |
| | 8B | Interior | | 14 |
| | 9A | Surface | | 20 |
| | 9F | Surface | | 16 |
| NR-74B-QA | 9D | Interior | 1350 | 12 |
| | 7C | Interior | | 12 |
| | 9B | Interior | | 16 |
| | 10A | Surface | | 24 |
| | 7A | Surface | | 20 |
| NR-74B-QA | 10D | Interior | 1500 | 13 |
| | 7D | Interior | | 13 |
| | 10B | Interior | | 16 |
| | 10F | Surface | | 25 |
| | 8A | Surface | | 29 |

¹Heat treatment:

NR-74B-F: As-forged; 16 hr at 1400° F.

NR-74B-QA: 2300° F water-quenched; 16 hr at 1400° F.



TABLE III

RUPTURE TEST DATA AT 1200°, 1350°, and 1500° F FOR 8-590 ALLOY DISCS NR-74B

| Disc (a) | Specimen number | Specimen location (b) | Test temperature (°F) | Stress (psi) | Rupture time (hr) | Elongation in 1 in. (percent) | Reduction of area (percent) | Minimum creep rate (percent/hr) | | |
|------------|-----------------|-----------------------|-----------------------|--------------|-------------------|-------------------------------|-----------------------------|---------------------------------|------|-------|
| °NR-74B-F | 17Y | CRR | 1200 | 55,000 | 69.5 | 21 | 20.6 | ----- | | |
| | 17Y | CRR | | 50,000 | 150 | 17 | 21.2 | ----- | | |
| | 17Z | SRR | | d50,000 | e1288 | ----- | ----- | ----- | | |
| | 17Y | CRR | | 45,000 | 372.5 | f9 | 14.4 | 0.0102 | | |
| | 12Y | CRR | | 40,000 | 894 | 6 | 7.3 | .0036 | | |
| | 17Z | SRR | | d40,000 | 1396 | 7 | 7.1 | .0018 | | |
| | 12Y | CRR | | 37,000 | 2310 | 7 | 12.7 | .0013 | | |
| | 12X | SRR | | d37,000 | e2376 | ----- | ----- | .0012 | | |
| | 17X | SRR | | 52,500 | 256 | f15 | 15.0 | ----- | | |
| | 12Y-C | CRC | | 52,500 | 161 | 17 | 15.6 | ----- | | |
| | °NR-74B-Q | 19Y | | CRR | 1200 | 55,000 | 59 | 4 | 7.9 | ----- |
| | | 19Y | | CRR | | 50,000 | 74 | 6 | 8.5 | ----- |
| 19Y | | CRR | 45,000 | 493 | | f4 | 7.3 | .0046 | | |
| 14Y | | CRR | 42,000 | 495 | | f5 | 8.5 | .0068 | | |
| 18Y | | CRR | 40,000 | 937 | | f5 | 5.0 | .0026 | | |
| 19X | | SRR | 52,500 | 14 | | 6 | 13.6 | ----- | | |
| 19X | | SRR | 50,000 | 111 | | 4 | 10.9 | ----- | | |
| 14YC | | CRC | 50,000 | 228 | | f3 | 15.3 | ----- | | |
| 17Z | | SRR | 50,000 | 10 | | f8 | 6.2 | ----- | | |
| °NR-74B-QA | | 13Y | CRR | 1200 | | 55,000 | 60 | 11 | 13.8 | ----- |
| | 20Y | CRR | 50,000 | | 153 | 13 | 16.7 | ----- | | |
| | 20Y | CRR | 45,000 | | 640 | f9 | 13.3 | .0086 | | |
| | 13Y | CRR | 42,000 | | 878 | 12 | 13.3 | .0072 | | |
| | 13X | SRR | d40,000 | | 1596 | 13 | 19.1 | .0036 | | |
| | 20X | SRR | 52,000 | | 95 | 13 | 17.8 | ----- | | |
| °NR-74B-F | 17Y | CRR | 1350 | 30,000 | 60 | 12 | 14.4 | ----- | | |
| | 17Y | CRR | | 25,000 | 180 | 8 | 11.5 | .0186 | | |
| | 12X | SRR | | d25,000 | 183 | 11 | 10.9 | .0186 | | |
| | 17Y | CRR | | 20,000 | 676 | 7 | 4.4 | .0030 | | |
| | 12X | SRR | | d20,000 | f196 | f3.5 | 6.4 | .0054 | | |
| | 12Y | CRR | | 17,000 | 1291 | 4 | 2.3 | .0006 | | |
| | 17Z | SRR | | d17,000 | 995 | f5 | 2.4 | .0006 | | |
| | 17X | SRR | | 27,500 | 86 | 6 | 10.9 | ----- | | |
| | 12Y-C | CRC | | 27,500 | 198 | 17.5 | 18.9 | ----- | | |
| | °NR-74B-Q | 19Y | | CRR | 1350 | 33,000 | 86 | 10 | 15.6 | ----- |
| 19Y | | CRR | 30,000 | 252 | | 11 | 15.0 | ----- | | |
| 14Y | | CRR | 27,000 | 204 | | 8 | 8.0 | ----- | | |
| 14Y | | CRR | 25,000 | 951 | | 11 | 17.8 | .0048 | | |
| 14X | | SRR | 32,000 | 165.5 | | f4 | 5.0 | ----- | | |
| 14Y-C | CRC | 32,000 | 373 | 5 | 7.3 | ----- | | | | |
| °NR-74B-QA | 13Y | CRR | 1350 | 35,000 | 45.5 | 13 | 17.8 | ----- | | |
| | 13Y | CRR | | 30,000 | 198 | 12.5 | 16.0 | ----- | | |
| | 13Y | CRR | | 27,000 | 167 | 8 | 15.6 | ----- | | |
| | 13Y | CRR | | 25,000 | 1121 | f13 | 17.8 | .0044 | | |
| | 13X | SRR | | d25,000 | e844 | ----- | ----- | .0050 | | |
| | 20X | SRR | | 32,000 | 97 | f18 | 18.3 | ----- | | |
| °NR-74B-F | 9A | SRR | 1500 | 20,000 | 29 | 6.5 | 12.8 | .10 | | |
| | 9D | CRR | | 16,000 | 59 | 9.0 | 8.5 | .05 | | |
| | 9E | CRR | | 11,000 | 124 | 12.0 | 17.5 | .024 | | |
| | 2Z | SRR | | h10,000 | 264 | 10.0 | ----- | .0083 | | |
| | 9F | SRR | | 6,000 | 1018 | 5.0 | 9.4 | .0018 | | |
| °NR-74B-QA | 11A | SRR | 1500 | 20,000 | 76 | 18.0 | 25.5 | ----- | | |
| | 11F | SRR | | 20,000 | 104 | 30.0 | 33.9 | .027 | | |
| | 11E | CRR | | 19,000 | 203 | 32.0 | 33.6 | ----- | | |
| | 11C | CRR | | 18,000 | 372 | 27.0 | 28.8 | .0065 | | |
| | 11B | CRR | | 16,000 | 642 | 20.0 | 31.6 | .0038 | | |
| | 11D | CRR | | 15,000 | 1000 | 16.5 | 25.5 | .0014 | | |

°Heat treatments:

NR-74B-F As-forged; 16 hr at 1400° F.

NR-74B-Q 2300° F water-quenched.

NR-74B-QA 2300° F water-quenched; 16 hr at 1400° F.

bCRR center-plane radial specimen near rim of disc.
 SRR surface-plane radial specimen near rim of disc.
 CRC center-plane radial specimen near center of disc.
 SRR surface-plane tangential specimen near rim of disc.

°NACA data. (Specimens were 0.160 in. in diameter with a gage length of 1 in. unless indicated otherwise.)

dTest on 0.250-in.-diameter specimen with precision extensometers.

eDiscontinued at this time.

fFractured in gage mark.

hNDEP and Navy data. (Specimens were 0.250 in. in diameter with gage length of 1.3 in.)

hTest on 0.505-in.-diameter specimen.



TABLE IV

RUPTURE TEST CHARACTERISTICS AT 1200°, 1350°, AND 1500° F OF S-590 ALLOY DISCS NR-74B

| Disc (a) | Temperature (°F) | Rupture strength | | | | Rupture ductility | | | |
|------------------------|---------------------|-------------------------------|--------|---------|---------|---|--------|---------|---------|
| | | Stress (psi) for rupture in - | | | | Estimated elongation (percent) to rupture in - | | | |
| | | 10 hr | 100 hr | 1000 hr | 2000 hr | 10 hr | 100 hr | 1000 hr | 2000 hr |
| ^b NR-74B-F | 1200 | ^c 69,000 | 52,500 | 40,000 | 37,000 | -- | 20 | 7 | 7 |
| ^b NR-74B-Q | | ^c 66,000 | 51,000 | 40,000 | 37,000 | -- | 6 | 5 | -- |
| ^b NR-74B-QA | | ^c 66,000 | 52,000 | 42,000 | 38,500 | -- | 12 | 12 | 12 |
| ^b NR-74B-F | 1350 | ^c 42,000 | 27,500 | 18,000 | 16,000 | -- | 10 | 5 | -- |
| ^b NR-74B-Q | | ^c 42,000 | 32,500 | 25,000 | 23,000 | -- | 10 | 11 | -- |
| ^b NR-74B-QA | | ^c 41,000 | 32,000 | 25,000 | 23,500 | 15 | 12 | 13 | -- |
| ^d NR-74B-F | 1500 | 29,000 | 13,100 | 6,000 | 4,800 | 7 | 12 | 5 | -- |
| ^d NR-74B-QA | | ----- | 20,000 | 15,000 | 13,100 | -- | 30 | 16 | -- |

^aHeat treatments:

NR-74B-F As-forged; 16 hr at 1400° F.

NR-74B-Q 2300° F water-quenched.

NR-74B-QA 2300° F water-quenched; 16 hr at 1400° F.

^bNACA data.

^cEstimated strength by extrapolation.

^dNDRC and Navy data.



TABLE V

DATA ON STRESS AND TIME FOR TOTAL DEFORMATION AT 1200° F FOR S-590 ALLOY DISCS NR-74B

[NACA data]

| Disc (a) | Specimen number | Stress (psi) | Initial deformation (percent) | Time (hr) for total deformations of - | | | | | | Transition to third-stage creep | |
|-------------|--------------------|-----------------|-------------------------------------|---------------------------------------|-------------|-------------------|-----------|-----------|-----------|------------------------------------|--------------------------|
| | | | | 0.1 percent | 0.2 percent | 0.5 percent | 1 percent | 2 percent | 5 percent | Time (hr) | Deformation (percent) |
| NR-74B-F | 13Y | 25,000 | 0.107 | -- | 320 | ----- | --- | ---- | ---- | ---- | --- |
| | 15Y | 25,000 | .103 | -- | 270 | ----- | --- | ---- | ---- | ---- | --- |
| | 13X | 35,000 | .150 | -- | 4 | 612 | --- | ---- | ---- | ---- | --- |
| | 12Y | 37,000 | .165 | -- | --- | 6.5 | 286 | 1150 | 2265 | 1720 | 2.8 |
| | 12X | 37,000 | .190 | -- | 2 | 122 | 610 | 1740 | ---- | ---- | --- |
| | 12Y | 40,000 | .180 | -- | --- | 21.0 | 124 | 407 | 835 | 560 | 2.7 |
| | 17Z | 40,000 | .220 | -- | --- | 23 | 236 | 790 | ---- | 790 | 2.0 |
| | 17Y | 45,000 | .205 | -- | --- | 11 | 33 | 111 | 277 | 140 | 2.3 |
| | 17Y | 50,000 | .230 | -- | --- | 3 | 13 | 35 | 96 | 50 | 2.6 |
| 17Y | 55,000 | .260 | -- | --- | ----- | --- | 16 | 37 | ---- | --- | |
| NR-74B-Q | 15X | 35,000 | .199 | -- | --- | 187 | 590 | ---- | ---- | ---- | --- |
| | 18Y | 40,000 | .250 | -- | --- | 25 | 168 | 390 | ---- | 890 | 3.5 |
| | 14Y | 42,000 | .310 | -- | --- | 4 | 48 | 200 | ---- | 340 | 3.0 |
| | 19Y | 45,000 | .440 | -- | --- | 0.5 | 27 | 185 | ---- | 480 | 3.6 |
| | 19Y | 50,000 | .800 | -- | --- | ----- | --- | 20 | ---- | ---- | --- |
| NR-74B-QA | 13Y | 25,000 | .108 | -- | 130 | ^b 1250 | --- | ---- | ---- | ---- | --- |
| | 13Z | 35,000 | .158 | -- | 5 | 102 | 583 | ---- | ---- | ---- | --- |
| | 13X | 40,000 | .240 | -- | --- | 13 | 55 | 235 | 925 | 790 | 4.2 |
| | 13Y | 42,000 | .215 | -- | --- | 7 | 29 | 125 | 525 | 505 | 4.8 |
| | 20Y | 45,000 | .270 | -- | --- | 2 | 20 | 82 | 390 | 405 | 5.1 |
| | 20Y | 50,000 | .430 | -- | --- | ----- | 3 | 17 | 80 | 80 | 5.0 |

^aHeat treatments:

NR-74B-F As-forged; 16 hr at 1400° F.

NR-74B-Q 2300° F water-quenched.

NR-74B-QA 2300° F water-quenched; 16 hr at 1400° F.

^bBy extrapolation.

NACA

TABLE VI

DATA ON STRESS AND TIME FOR TOTAL DEFORMATION AT 1350° F FOR S-590 ALLOY DISCS NR-74B

| Disc (a) | Specimen number | Stress (psi) | Initial deformation (percent) | Time (hr) for total deformations of-- | | | | | | Transition to third-stage creep | |
|-----------|-----------------|--------------|-------------------------------|---------------------------------------|-------------|-------------------|-----------|-----------|-----------|---------------------------------|-----------------------|
| | | | | 0.1 percent | 0.2 percent | 0.5 percent | 1 percent | 2 percent | 5 percent | Time (hr) | Deformation (percent) |
| NR-74B-F | b1Y | 12,000 | 0.052 | 13 | 158 | 1530 | ----- | ----- | --- | --- | --- |
| | b2X | 15,000 | .092 | 5 | 78 | 522 | 1282 | 1825 | --- | 850 | 0.66 |
| | c12Y | 17,000 | .085 | -- | 6 | 44 | 480 | 920 | --- | 600 | 1.2 |
| | c17Z | 17,000 | .089 | -- | 6 | 100 | 510 | 850 | --- | 600 | 1.2 |
| | b3X | 20,000 | .099 | 1 | 10 | 71 | 285 | 590 | --- | 375 | 1.28 |
| | c17Y | 20,000 | .100 | -- | ----- | d ₂ | 20 | 260 | 655 | 260 | 2.0 |
| | c12X | 20,000 | .115 | -- | ----- | 4 | 65 | ----- | --- | --- | ----- |
| | c17Y | 25,000 | .130 | -- | ----- | ----- | 11 | 58 | 150 | 65 | 2.1 |
| | c12X | 25,000 | .154 | -- | ----- | ----- | 13 | 68 | 160 | 85 | 2.4 |
| | c17Y | 30,000 | .160 | -- | ----- | ----- | 3 | 14 | 42 | --- | ----- |
| NR-74B-Q | c14Y | 25,000 | .115 | -- | 2.5 | d ₉ | 28 | 72 | 685 | 740 | 5.2 |
| | c14Y | 27,000 | .125 | -- | ----- | d ₂ | 9 | 50 | 170 | 170 | 5 |
| | c19Y | 30,000 | .145 | -- | ----- | d ₄ | 15 | 50 | 190 | 150 | 4 |
| | c19Y | 33,000 | .175 | -- | ----- | d ₄ | 13 | 31 | 65 | --- | ----- |
| NR-74B-QA | b2Y | 12,000 | .049 | 22 | 275 | ----- | ----- | ----- | --- | --- | ----- |
| | b1X | 15,000 | .085 | 6 | 58 | d ₃₀₃₀ | ----- | ----- | --- | --- | ----- |
| | b3X | 20,000 | e.125 | -- | 12 | 103 | 1765 | ----- | --- | --- | ----- |
| | b3Z | 23,000 | .142 | -- | 1.5 | 42 | 235 | ----- | --- | 460 | 1.35 |
| | c13Y | 25,000 | .110 | -- | 2 | 16 | 52 | 220 | 725 | 600 | 3.7 |
| | c13X | 25,000 | .142 | -- | 2 | 11 | 46 | 186 | ----- | --- | ----- |
| | c13Y | 27,000 | .120 | -- | ----- | 4 | 11 | 30 | 142 | 98 | 3.7 |
| | c13Y | 30,000 | .135 | -- | ----- | 6 | 14 | 32 | 133 | 90 | 3.7 |

^aHeat treatments:

NR-74B-F As-forged; 16 hr at 1400° F.

NR-74B-Q 2300° F water-quenched.

NR-74B-QA 2300° F water-quenched; 16 hr at 1400° F.

^bNDRC and Navy data.^cNACA data.^dEstimated.^eContraction upon release of load.

NACA

TABLE VII

DATA ON STRESS AND TIME FOR TOTAL DEFORMATION AT 1500° F FOR S-590 ALLOY DISCS NR-74B

[NDRC and Navy data]

| Disc (a) | Specimen number | Stress (psi) | Initial deformation (percent) | Time (hr) for total deformations of- | | | | | | Transition to third-stage creep | |
|-------------|--------------------|-----------------|-------------------------------------|--------------------------------------|-------------------|-------------|-------------------|-----------|-----------|------------------------------------|--------------------------|
| | | | | 0.1 percent | 0.2 percent | 0.5 percent | 1 percent | 2 percent | 5 percent | Time (hr) | Deformation (percent) |
| NR-74B-F | 9A | 20,000 | ----- | --- | ----- | 1.7 | 6 | 16 | 28 | 15.5 | 1.95 |
| | 9D | 16,000 | ----- | --- | ----- | 4.5 | 13.5 | 31 | 54 | 24 | 1.56 |
| | 9E | 11,000 | ----- | --- | 2 | 11 | 28 | 64 | 101 | 48 | 1.42 |
| | 2Z | 10,000 | 0.069 | 1 | 5 | 35 | 86 | 145 | 250 | 74 | 0.92 |
| | 9F | 6,000 | ----- | 4 | 22 | 132 | 392 | 700 | --- | 464 | 1.17 |
| NR-74B-QA | 11F | 20,000 | ----- | --- | ----- | 10 | 24 | 42 | 69 | 14 | .60 |
| | 11C | 18,000 | ----- | --- | ----- | 20 | 72 | 119 | 220 | 40 | .62 |
| | 11B | 16,000 | ----- | --- | 6 | 70 | 180 | 306 | 474 | 155 | .85 |
| | 11D | 15,000 | ----- | --- | 5 | 58 | 325 | 545 | 787 | 300 | .94 |
| | 4X | 12,000 | .077 | 3 | 25 | 1270 | ^b 3400 | --- | --- | 1700 | .58 |
| | 1Z | 10,000 | .068 | 72 | 456 | 1800 | ----- | --- | --- | ----- | ----- |
| | 2X | 8,000 | .036 | 430 | ^b 4000 | ----- | ----- | --- | --- | ----- | ----- |

^aHeat treatments:

NR-74B-F As-forged; 16 hr at 1400° F.

NR-74B-QA 2300° F water-quenched; 16 hr at 1400° F.

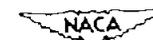
^bEstimated.

TABLE VIII
 TIME-DEFORMATION AND CREEP STRENGTHS AT 1200° F
 FOR S-590 ALLOY DISCS NR-74B

[NACA data]

| Disc (a) | Total deformation (percent) | Stress (psi) to cause total deformation in - | | | | | Creep strength (based on creep rates at 1000 hr) (psi) | |
|-----------|-----------------------------|--|--------|---------------------|---------------------|---------------------|--|--------------------|
| | | 1 hr | 10 hr | 100 hr | 1000 hr | 2000 hr | 0.00010 percent/hr | 0.00001 percent/hr |
| NR-74B-F | 0.2 | 38,000 | 33,000 | 28,500 | 22,000 | ^b 20,500 | 27,500 | ----- |
| NR-74B-Q | .2 | ----- | ----- | ----- | ----- | ----- | ----- | ----- |
| NR-74B-QA | .2 | ^b 40,000 | 33,000 | 26,000 | ^b 18,500 | ----- | 23,000 | ----- |
| NR-74B-F | .5 | ----- | 44,000 | 38,000 | 32,000 | ^b 30,000 | ----- | ----- |
| NR-74B-Q | .5 | ----- | 41,500 | 36,200 | 31,000 | ----- | ----- | ----- |
| NR-74B-QA | .5 | 47,000 | 40,500 | 33,800 | 27,000 | ^b 25,000 | ----- | ----- |
| NR-74B-F | 1.0 | ----- | 50,000 | 42,000 | 34,300 | 32,000 | ----- | ----- |
| NR-74B-Q | 1.0 | ----- | 47,000 | 40,500 | 34,000 | ^b 32,000 | ----- | ----- |
| NR-74B-QA | 1.0 | ----- | 46,500 | 39,500 | 33,000 | ^b 31,000 | ----- | ----- |
| NR-74B-F | Transition | ----- | ----- | 47,000 | 39,000 | 36,000 | ----- | ----- |
| NR-74B-Q | Transition | ----- | ----- | ^b 50,000 | 39,000 | ----- | ----- | ----- |
| NR-74B-QA | Transition | ----- | ----- | 49,000 | 39,000 | 36,000 | ----- | ----- |

^aHeat treatments:

NR-74B-F As-forged; 16 hr at 1400° F.

NR-74B-Q 2300° F water-quenched.

NR-74B-QA 2300° F water-quenched; 16 hr at 1400° F.

^bEstimated strength by extrapolation.

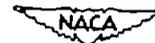


TABLE IX

TIME-DEFORMATION AND CREEP STRENGTHS AT 1350° F

FOR S-590 ALLOY DISCS NR-74B

[NACA, NDRC, and Navy data]

| Disc (a) | Total deformation (percent) | Stress (psi) to cause total deformation in - | | | | | Creep strength (based on minimum rates) (psi) | |
|-------------|-----------------------------------|---|---------------------|--------|--------------------|--------------------|---|--------------------|
| | | 1 hr | 10 hr | 100 hr | 1000 hr | 2000 hr | 0.00010 percent/hr | 0.00001 percent/hr |
| NR-74B-F | 0.1 | 20,000 | 12,800 | ----- | ----- | ----- | 10,600 | ----- |
| NR-74B-Q | .1 | ----- | ----- | ----- | ----- | ----- | ----- | ----- |
| NR-74B-QA | .1 | 19,000 | 13,800 | ----- | ----- | ----- | 16,400 | 12,100 |
| NR-74B-F | .2 | ----- | 18,600 | 13,100 | ^b 8,000 | ----- | ----- | ----- |
| NR-74B-Q | .2 | ----- | ^b 16,000 | ----- | ----- | ----- | ----- | ----- |
| NR-74B-QA | .2 | 26,400 | 20,600 | 14,700 | 8,700 | ^b 7,000 | ----- | ----- |
| NR-74B-F | .5 | ----- | 21,000 | 17,000 | 13,000 | 11,500 | ----- | ----- |
| NR-74B-Q | .5 | ----- | 24,000 | ----- | ----- | ----- | ----- | ----- |
| NR-74B-QA | .5 | 30,000 | 25,700 | 21,400 | 17,000 | 15,700 | ----- | ----- |
| NR-74B-F | 1.0 | ----- | 25,500 | 20,500 | 15,500 | 14,000 | ----- | ----- |
| NR-74B-Q | 1.0 | ----- | 33,000 | ----- | ----- | ----- | ----- | ----- |
| NR-74B-QA | 1.0 | ----- | 27,700 | 24,100 | 20,800 | 19,800 | ----- | ----- |
| NR-74B-F | Transition | ----- | ----- | 24,500 | 14,500 | 11,700 | ----- | ----- |
| NR-74B-Q | Transition | ----- | ----- | 31,000 | 14,000 | ----- | ----- | ----- |
| NR-74B-QA | Transition | ----- | ----- | 29,000 | 22,500 | ----- | ----- | ----- |

^aHeat treatments:

NR-74B-F As-forged; 16 hr at 1400° F.

NR-74B-Q 2300° F water-quenched.

NR-74B-QA 2300° F water-quenched; 16 hr at 1400° F.

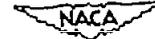
^bEstimated strength by extrapolation.

TABLE X
 TIME-DEFORMATION AND CREEP STRENGTHS AT 1500° F
 FOR S-590 ALLOY DISCS NR-74B
 [NDRC and Navy data]

| Disc (a) | Total deformation (percent) | Stress (psi) to cause total deformation in - | | | | | Creep strength (based on minimum rates) (psi) | |
|-------------|-----------------------------------|---|--------------------|--------|--------------------|---------|---|--------------------|
| | | 1 hr | 10 hr | 100 hr | 1000 hr | 2000 hr | 0.00010 percent/hr | 0.00001 percent/hr |
| NR-74B-F | 0.1 | 10,000 | ^b 6,800 | ----- | ----- | ----- | ^b 2,800 | ---- |
| NR-74B-QA | .1 | ^b 13,600 | 11,500 | 9,400 | ^b 7,300 | ----- | 10,000 | 7100 |
| NR-74B-F | .2 | 13,500 | 7,800 | ----- | ----- | ----- | ----- | ---- |
| NR-74B-QA | .2 | ----- | 14,000 | 11,000 | 9,200 | 8,600 | ----- | ---- |
| NR-74B-F | .5 | ----- | 12,700 | 6,500 | ----- | ----- | ----- | ---- |
| NR-74B-QA | .5 | ----- | 19,400 | 14,800 | 11,600 | 10,500 | ----- | ---- |
| NR-74B-F | 1.0 | ----- | 17,300 | 9,000 | ^b 4,000 | ----- | ----- | ---- |
| NR-74B-QA | 1.0 | ----- | ----- | 17,200 | 13,600 | 12,700 | ----- | ---- |
| NR-74B-F | Transition | ----- | ----- | 9,300 | ^b 4,200 | ----- | ----- | ---- |
| NR-74B-QA | Transition | ----- | 20,400 | 16,700 | 12,800 | 11,800 | ----- | ---- |

^aHeat treatments:

NR-74B-F As-forged; 16 hr at 1400° F.

NR-74B-QA 2300° F water-quenched; 16 hr at 1400° F.

^bEstimated strength by extrapolation.

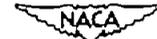


TABLE XI

CREEP TEST DATA AT 1200°, 1350°, AND 1500° F FOR S-590 ALLOY DISCS NR-74B

24

| Disc (a) | Specimen number | Test temperature (°F) | Stress (psi) | Duration (hr) | Deformation upon application of load (percent) | Creep rate (percent/hr) at - | | | | Total deformation (percent) at - | | | |
|-----------|------------------|-----------------------|--------------|---------------|--|------------------------------|----------------------|---------|---------------------|----------------------------------|--------------------|---------|-------------------|
| | | | | | | 500 hr | 1000 hr | 1500 hr | 2000 hr | 500 hr | 1000 hr | 1500 hr | 2000 hr |
| NR-74B-F | b _{13Y} | 1200 | 25,000 | 1108 | 0.107 | 0.000082 | 0.000066 | ----- | ----- | 0.217 | 0.257 | ----- | ----- |
| | b _{15Y} | 1200 | 25,000 | 960 | .103 | .000082 | .000066 | ----- | ----- | .217 | ----- | ----- | ----- |
| | b _{13X} | 1200 | 35,000 | 1002 | .150 | .00028 | .00027 | ----- | ----- | .469 | .605 | ----- | ----- |
| NR-74B-Q | b _{15X} | 1200 | 35,000 | 770 | .199 | .00095 | ^c 0.00090 | ----- | ----- | .915 | ^c 1.163 | ----- | ----- |
| NR-74B-QA | b _{13Y} | 1200 | 25,000 | 1009 | .108 | .00026 | .00025 | ----- | ----- | .305 | .435 | ----- | ----- |
| | b _{13Z} | 1200 | 35,000 | 1002 | .158 | .00090 | .00087 | ----- | ----- | .925 | 1.370 | ----- | ----- |
| NR-74B-F | d _{3X} | 1350 | 20,000 | (e) | .099 | ^f 0.0032 | ----- | ----- | ----- | 1.65 | ----- | ----- | ----- |
| | d _{2X} | 1350 | 15,000 | 1872 | .092 | .00064 | .00058 | 0.0013 | ^g 0.0025 | .483 | .760 | 1.22 | ^g 2.00 |
| | d _{1Y} | 1350 | 12,000 | 2059 | .052 | .00022 | .00019 | .00018 | .00017 | .301 | .405 | .497 | .586 |
| NR-74B-QA | d _{3Z} | 1350 | 23,000 | 4886 | .142 | .0015 | ----- | ----- | ----- | 1.42 | ----- | ----- | ----- |
| | d _{3X} | 1350 | 20,000 | 2016 | .125 | .00026 | .00013 | .00012 | .00012 | .809 | .907 | .969 | 1.029 |
| | d _{1X} | 1350 | 15,000 | 2282 | .085 | .00011 | .00008 | .00008 | .00005 | .323 | .385 | .420 | .440 |
| | d _{2Y} | 1350 | 12,000 | 2135 | .049 | .00007 | .000019 | .000019 | .000009 | .219 | .237 | .246 | .253 |
| NR-74B-F | d _{1Y} | 1500 | 8,000 | 1743 | .046 | ⁱ 0.00185 | ----- | ----- | ----- | 1.021 | ----- | ----- | ----- |
| NR-74B-QA | d _{4X} | 1500 | 12,000 | 2136 | .077 | .00020 | .00017 | .00017 | .00020 | .375 | .457 | .536 | .640 |
| | d _{1Z} | 1500 | 10,000 | 2064 | .068 | .00027 | .00022 | .00022 | .00025 | .228 | .330 | .430 | .620 |
| | d _{2X} | 1500 | 8,000 | 2039 | .036 | .000044 | .000034 | .000028 | .000024 | .104 | .125 | .140 | .152 |

^aHeat treatments:

NR-74B-F As-forged; 16 hr at 1400° F.

NR-74B-Q 2300° F water-quenched.

NR-74B-QA 2300° F water-quenched; 16 hr at 1400° F.

^bNACA data.^cAt 770 hr.^dNDRC and Navy data.^eBroke in threads shortly after 744 hr.^fMinimum creep rate, measured between 75 and 400 hr; 0.00255 percent/hr.^gAt 1872 hr.^hDiscontinued at 886 hr with 2.78-percent deformation. Minimum creep rate 0.0015 percent/hr between 250 and 500 hr.ⁱDiscontinued at 743 hr with 1.56-percent deformation. Minimum creep rate between 150 and 350 hr; 0.00166 percent/hr.

Data from this test were not used for the design curves.

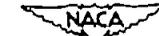


TABLE XII

EFFECT OF CREEP TESTING ON THE ROOM-TEMPERATURE PHYSICAL PROPERTIES OF 8-590 ALLOY DISCS NR-74B

| Disc (a) | Specimen number | Prior testing conditions | | | Residual room-temperature properties | | | | | | | | |
|-----------------|----------------------|-----------------------------|-----------------|--------------|--------------------------------------|-----------------------------------|-------------|-------------|--------------------------------|-------------------------------------|-----------------------------------|----------------------------------|---------------------|
| | | Temper- ature (°F) | Stress (psi) | Time (hr) | Tensile strength (psi) | Offset yield strength (psi) | | | Proportional limit (psi) | Elongation in 2 in. (percent) | Reduction of area (percent) | Izod impact strength (psi) | Vickers hardness |
| | | | | | | 0.02 percent | 0.1 percent | 0.2 percent | | | | | |
| NR-74B-F | (b) | (c) | (c) | (c) | 129,050 | 70,750 | 90,000 | 98,250 | 42,500 | 8 | 10.3 | ----- | 309 |
| | d _{14Y} | (o) | (o) | (o) | ----- | ----- | ----- | ----- | ----- | ----- | ----- | e ₂ , 5 | --- |
| | f _{3Y} , 4X | (o) | (o) | (o) | ----- | ----- | ----- | ----- | ----- | ----- | ----- | e ₇ , 6, 5 | 268 |
| | d _{13Y} | 1200 | 25,000 | 1108 | 127,000 | 61,000 | 85,000 | 94,500 | 27,500 | 6 | 7.3 | ----- | --- |
| | d _{13X} | 1200 | 35,000 | 1002 | ----- | ----- | ----- | ----- | ----- | ----- | ----- | e ₂ , 2 | 278 |
| | f _{1Y} | 1350 | 12,000 | 2059 | 110,500 | 58,000 | 76,000 | 85,000 | 40,500 | 7 | 1.4 | ----- | --- |
| | f _{2X} | 1350 | 15,000 | 1872 | ----- | ----- | ----- | ----- | ----- | ----- | ----- | e ₁ , 2 | 245 |
| | f _{1X} | 1500 | 8,000 | 743 | 105,000 | 53,500 | 71,200 | 80,800 | 37,900 | 1.5 | 1.6 | ----- | --- |
| (h) | 1500 | ----- | ----- | ----- | ----- | ----- | ----- | ----- | ----- | ----- | ----- | --- | |
| NR-74B-Q | (b) | (o) | (o) | (o) | 119,500 | 36,250 | 50,750 | 57,000 | 22,500 | 36 | 34.3 | ----- | 235 |
| | d _{14X} | (o) | (o) | (o) | ----- | ----- | ----- | ----- | ----- | ----- | ----- | e ₂₄ , 32 | --- |
| | d _{15X} | 1200 | 35,000 | 770 | 127,500 | 58,000 | 81,000 | 87,500 | 25,000 | 6.5 | 5.7 | ----- | --- |
| (h) | 1200 | ----- | ----- | ----- | ----- | ----- | ----- | ----- | ----- | ----- | ----- | --- | |
| NR-74B-QA | (b) | (o) | (o) | (o) | 130,500 | 45,000 | 63,500 | 70,500 | 25,000 | 17 | 18.2 | ----- | 267 |
| | d _{16Y} | (o) | (o) | (o) | ----- | ----- | ----- | ----- | ----- | ----- | ----- | e ₉ , 28 | --- |
| | f _{3Y} , 4Z | (o) | (o) | (o) | ----- | ----- | ----- | ----- | ----- | ----- | ----- | e ₇ , 8, 6 | 282 |
| | d _{13Y} | 1200 | 25,000 | 1009 | 131,000 | 60,000 | 78,000 | 85,000 | 37,500 | 5.5 | 6.4 | ----- | --- |
| | d _{13Z} | 1200 | 35,000 | 1002 | ----- | ----- | ----- | ----- | ----- | ----- | ----- | e ₅ , 4 | 284 |
| | f _{2Y} | 1350 | 12,000 | 2135 | 132,500 | 57,500 | 72,000 | 79,600 | 39,000 | 3.3 | 4.5 | ----- | --- |
| | f _{1X} | 1350 | 15,000 | 2282 | ----- | ----- | ----- | ----- | ----- | ----- | ----- | e ₄ , 4 | 319 |
| | f _{1Z} | 1500 | 10,000 | 2064 | 116,000 | 43,000 | 55,000 | 62,500 | 31,500 | 4.5 | 4.9 | ----- | --- |
| f _{4X} | 1500 | 12,000 | 2136 | ----- | ----- | ----- | ----- | ----- | ----- | ----- | e ₂ , 3 | 295 | |

^aHeat treatments:

NR-74B-F As forged; 16 hr at 1400° F.

NR-74B-Q 2300° F water-quenched.

NR-74B-QA 2300° F water-quenched; 16 hr at 1400° F.

^bAverage of tests on center- and surface-planes radial specimens.^cOriginal condition.^dNACA data.^eSpecimens were 0.365-in. square with a 0.50-in.-deep V-notch.^fNDSC and Navy data.^gSpecimens were 0.450-in.-diameter, V-notch.^hNo specimen available for impact and hardness tests.

NACA

TABLE XIII
 RESULTS FROM ALLEGHENY-INDIUM STEEL CORPORATION AND COMPARATIVE NACA, UMBC, AND NAVY
 RESULTS ON THE S-590 ALLOY DISCS HB-74B

| Data source (1) | Treatment | Room-temperature tensile properties | | | | | Rupture properties | | | | | |
|--------------------------------|---|-------------------------------------|--|----------------------|-----------------------------|------------------|--------------------|------------------|--------------|----------------------|-----------------------------|------------------|
| | | Tensile strength (psi) | 0.02-percent-offset yield strength (psi) | Elongation (percent) | Reduction of area (percent) | Brinell hardness | Temperature (°F) | Stress (psi) | Time (hr) | Elongation (percent) | Reduction of area (percent) | Brinell hardness |
| As-forged disc | | | | | | | | | | | | |
| AL | As-forged | 143,000 141,500 140,500 | 92,500 85,000 82,500 | 3.5 12.5 7 | 7.7 29.8 13.1 | 293-311 | 1350 | 30,000 | 25 | 6 | 18.4 | 293 |
| AL | As-forged; 16 hr at 1400° F | 133,500 | 72,500 | 13 | 19.2 | 302 | 1200 1200 | 50,000 45,000 | 134.5 397 | 17.5 13 | 30 16 | 297 271 |
| UM | | 129,050 | 70,750 | 8 | 10.3 | 267-311 | 1200 1200 | 50,000 45,000 | 150 372.5 | 17 9 | 21.2 14.4 | --- |
| AL | Forged; 16 hr at 1500° F | 132,500 | 75,000 | 13 | 10.1 | 285 | 1500 1500 | 20,000 17,500 | 9 23 | 15 23 | 27 22 | 277 285 |
| Solution-treated disc | | | | | | | | | | | | |
| AL | 2300° F 3 $\frac{1}{4}$ hr water-quenched | 125,000 | 32,500 | 39.5 | 34.3 | 229-241 | 1200 | 45,000 | 178 | 3 | 4 | 223 |
| UM | | 119,500 | 36,250 | 36 | 34.3 | 211-223 | 1200 | 45,000 | 493 | 4 | 7.3 | --- |
| Solution-treated and aged disc | | | | | | | | | | | | |
| AL | 2300° F water- quenched; 16 hr at 1400° F | 134,000 | 45,000 | 22 | 18.8 | 255 | 1350 | 35,000 | 42 | 24 | 23 | 302 |
| | | | | | | | 1350 | 30,000 | 185 | 26.5 | 29.4 | 302 |
| | | | | | | | 1350 | 25,000 | 983 | 19 | 23 | 302 |
| UM | | 130,500 | 45,000 | 17 | 18.2 | 259 | 1350 | 30,000 | 252 | 11 | 15 | --- |
| | | | | | | | 1350 | 25,000 | 951 | 11 | 17.8 | --- |
| B | | ----- | ----- | ----- | ----- | ----- | 1500 | 20,000 | 76 | 18 | 25.5 | --- |
| | | | | | | | 1500 | 15,000 | 1000 | 16.5 | 25.5 | --- |
| AL | 2300° F water- quenched; 16 hr at 1900° F | ----- | ----- | ----- | ----- | ----- | 1500 | 20,000 | 104 | 29.5 | 41.5 | 277 |
| | | | | | | | 1500 | 17,500 | 238 | 36 | 38 | 285 |
| | | | | | | | 1500 | 15,000 | 747 | 31 | 39 | 277 |

1 AL data supplied by Allegheny-Indium Steel Corp.
 UM University of Michigan (NACA) data.
 B Battelle (NEDC and Navy) data.

All Allegheny-Indium data were on specimens representing chords from the discs. All University of Michigan and Battelle data were on radial specimens.



NACA

TABLE XIV
COMPARATIVE PROPERTIES OF BAR STOCK AND DISCS OF S-590 ALLOY

| Form | Treatment | Tensile properties | | | | | Rupture characteristics | | | | | |
|------------------------|---------------------------|--------------------|------------------------------------|-----------------------------|-------------|----------------------|-------------------------|-------------------------------------|------------------|--|-----------|------------|
| | | Temperature (°F) | Tensile strength (psi) | Offset yield strength (psi) | | Elongation (percent) | Temperature (°F) | Rupture strength (psi) | | Estimated rupture elongation (percent) | | |
| | | | | 0.02 percent | 0.2 percent | | | 100 hr | 1000 hr | 100 hr | 1000 hr | |
| Disc (NR-74B-F) | Forged and aged | 75 | 129,050 | 70,750 | 98,250 | 8 | 1350 | 27,500 | 18,000 | 10 | 5 | |
| Disc (NR-74B-Q) | Solution-treated | 75 | 119,500 | 36,250 | 57,000 | 36 | 1350 | 32,500 | 25,000 | 10 | 11 | |
| Disc (NR-74B-QA) | Solution-treated and aged | 75 | 130,500 | 45,000 | 70,500 | 17 | 1350 | 32,000 | 25,000 | 12 | 13 | |
| Bar stock ^a | Solution-treated and aged | 75 | 159,500 | 56,750 | 88,250 | 9.5 | ^a 1350 | 31,000 32,000 | 24,000 26,000 | -- 35 | -- 40 | |
| Disc (NR-74B-F) | Forged and aged | 1350 | 64,625 | ----- | 55,000 | 29 | 1500 | 13,100 | 6,000 | 12 | 5 | |
| Disc (NR-74B-QA) | Solution-treated and aged | 1350 | 65,750 | ----- | 46,000 | 25 | 1500 | 20,000 | 15,000 | 30 | 16 | |
| Bar stock ^a | Solution-treated and aged | 1350 | 65,875 | ----- | 57,500 | 28 | ^b 1500 | 19,000 21,000 | 14,000 15,500 | 8 23 | 10 25 | |
| Form | Treatment | Temperature (°F) | Time-deformation strengths | | | | | | | | | |
| | | | 100-hr deformation strengths (psi) | | | | | 1000-hr deformation strengths (psi) | | | | |
| | | | 0.1 percent | 0.2 percent | 0.5 percent | 1 percent | Transition | 0.1 percent | 0.2 percent | 0.5 percent | 1 percent | Transition |
| Disc (NR-74B-F) | Forged and aged | 1500 | ----- | ----- | 6,500 | 9,000 | 9,300 | ----- | ----- | ----- | 4,000 | 4,200 |
| Disc (NR-74B-QA) | Solution-treated and aged | 1500 | 9,400 | 11,000 | 14,800 | 17,200 | 16,700 | 7,300 | 9,200 | 11,600 | 13,600 | 12,800 |
| Bar stock ^b | Solution-treated and aged | 1500 | 10,300 | 13,200 | 17,300 | 18,700 | 17,600 | 8,000 | 10,600 | 13,100 | 14,000 | ----- |

^aUnpublished data from the University of Michigan.

^bData from reference 11.



TABLE XV

COMPARISON OF ROOM-TEMPERATURE AND HIGH-TEMPERATURE PROPERTIES OF SEVERAL LARGE FORGED DISCS OF S-590 AND S-816 ALLOYS

| Test temperature, °F | Room temperature | | | | 1200 | | | | 1350 | | | | 1500 | | | |
|---|------------------|-----------|----------|----------|--------------------------------|---------------------|---------------------|---------------------|--------------------------------|---------------------|---------------------|---------------------|--------------------------------|--------------------|---------------------|---------------------|
| | S-590 | | S-816 | | S-590 | | S-816 | | S-590 | | S-816 | | S-590 | | S-816 | |
| | NR-74B-F | NR-74B-QA | NR-76B-F | NR-76B-Q | NR-74B-F | NR-74B-QA | NR-76B-F | NR-76B-Q | NR-74B-F | NR-74B-QA | NR-76B-F | NR-76B-Q | NR-74B-F | NR-74B-QA | NR-76B-F | NR-76B-Q |
| Short-time properties: | | | | | | | | | | | | | | | | |
| Charpy impact strength, ft-lb | 5 | 9 | 25 | 19 | 9 | 15 | 43 | 43 | 11 | 17 | 47 | 40 | 13 | 20 | 43 | 43 |
| Isod impact strength, ft-lb | 6 | 7 | 18 | 19 | | | | | | | | | | | | |
| Tensile strength, psi | 129,050 | 130,900 | 150,000 | 144,000 | 88,700 | 81,600 | 120,000 | 106,000 | 64,623 | 65,790 | 88,000 | 83,000 | 43,125 | 44,400 | 59,000 | 60,000 |
| 0.1-percent-offset yield strength, psi | 50,000 | 63,500 | 79,000 | 70,000 | 66,250 | 46,000 | 63,000 | 56,000 | 50,750 | 43,500 | 56,000 | 52,000 | 31,350 | 35,050 | 46,000 | 49,000 |
| 0.2-percent-offset yield strength, psi | 58,250 | 70,500 | 85,000 | 76,000 | 71,750 | 49,000 | 67,000 | 58,000 | 55,000 | 46,000 | 59,000 | 55,000 | 39,900 | 37,850 | 49,000 | 51,000 |
| Elongation, percent | 8 | 17 | 21 | 30 | 15 | 27 | 16 | 12 | 29 | 25 | 23 | 28 | 25 | 18 | 17 | 17 |
| Rupture strengths, psi: | | | | | | | | | | | | | | | | |
| 10-hr | ----- | ----- | ----- | ----- | ^o 69,000 | ^o 66,000 | ^o 78,000 | ^o 84,000 | ^o 42,000 | ^o 41,000 | 32,000 | ^o 53,000 | ^o 29,000 | ----- | ^o 31,000 | ^o 29,500 |
| 100-hr | ----- | ----- | ----- | ----- | 52,500 | 52,000 | 62,000 | 66,000 | 27,500 | 32,000 | 37,500 | 39,000 | 13,100 | 20,000 | 20,500 | 22,800 |
| 1000-hr | ----- | ----- | ----- | ----- | 40,000 | 42,000 | 50,000 | 53,000 | 18,000 | 29,000 | 27,000 | 29,000 | 6,000 | 15,000 | 13,700 | 17,500 |
| Rupture elongations, percent ^o : | | | | | | | | | | | | | | | | |
| 100-hr | ----- | ----- | ----- | ----- | 20 | 12 | 10 | 7 | 10 | 12 | 10 | 12 | 12 | 30 | 5 | 7 |
| 1000-hr | ----- | ----- | ----- | ----- | 7 | 12 | 10 | 7 | 5 | 13 | 10 | 10 | 5 | 16 | 4 | 4 |
| Creep strengths, psi: | | | | | | | | | | | | | | | | |
| 0.0001 percent/hr | ----- | ----- | ----- | ----- | 27,500 | 23,000 | 28,000 | 28,000 | 10,600 | 16,400 | 19,500 | 19,000 | ^o 2,800 | 10,000 | 11,000 | 13,500 |
| 0.0001 percent/hr | ----- | ----- | ----- | ----- | ----- | ----- | ^o 18,000 | ^o 16,000 | ----- | 12,100 | 13,000 | 10,500 | ----- | 7,100 | 8,500 | 7,500 |
| 100-hr deformation strengths, psi: | | | | | | | | | | | | | | | | |
| 0.1-percent deformation | ----- | ----- | ----- | ----- | ----- | ----- | ----- | ^o 19,500 | ----- | ----- | 13,000 | 13,000 | ----- | 9,400 | 9,000 | 9,000 |
| 0.2-percent deformation | ----- | ----- | ----- | ----- | 28,500 | 26,000 | 31,500 | 34,500 | 13,100 | 14,700 | 22,000 | 20,000 | ----- | 11,000 | 13,800 | 14,000 |
| 0.5-percent deformation | ----- | ----- | ----- | ----- | 38,000 | 33,800 | 43,000 | 46,000 | 17,000 | 21,400 | 27,000 | 27,000 | 6,500 | 14,800 | 17,000 | 19,000 |
| 1.0-percent deformation | ----- | ----- | ----- | ----- | 42,000 | 39,500 | 48,000 | 52,500 | 20,500 | 24,100 | 31,000 | 30,000 | 9,000 | 17,800 | 18,500 | 21,000 |
| Transition | ----- | ----- | ----- | ----- | 47,000 | 49,000 | 59,500 | 64,000 | 24,500 | 29,000 | 35,500 | 36,000 | 9,300 | 16,700 | 18,000 | 20,200 |
| 1000-hr deformation strengths, psi: | | | | | | | | | | | | | | | | |
| 0.1-percent deformation | ----- | ----- | ----- | ----- | ----- | ----- | ----- | ----- | ----- | ^o 10,000 | ^o 9,000 | ----- | ----- | ^o 7,300 | ^o 5,500 | ^o 5,500 |
| 0.2-percent deformation | ----- | ----- | ----- | ----- | 22,000 | ^o 18,500 | 24,500 | 25,500 | ^o 8,000 | 8,700 | 16,500 | 15,000 | ----- | 9,200 | 9,600 | 10,000 |
| 0.5-percent deformation | ----- | ----- | ----- | ----- | 32,000 | 27,000 | 33,500 | 37,000 | 13,000 | 17,000 | 24,500 | 21,500 | ----- | 11,600 | 11,200 | ^o 14,700 |
| 1.0-percent deformation | ----- | ----- | ----- | ----- | 34,300 | 33,000 | 38,000 | ^o 43,000 | 15,500 | 20,800 | ^o 26,500 | ^o 23,500 | ^o 4,000 | 13,600 | 12,000 | ^o 16,000 |
| Transition | ----- | ----- | ----- | ----- | 39,000 | 39,000 | 48,000 | 52,000 | 14,500 | 22,500 | 28,000 | 27,000 | ^o 4,200 | 12,800 | 12,000 | ^o 16,000 |
| Residual room-temperature properties: | | | | | | | | | | | | | | | | |
| Isod impact strength, ft-lb | ----- | ----- | ----- | ----- | After creep testing at 1200° F | | | | After creep testing at 1350° F | | | | After creep testing at 1500° F | | | |
| Tensile strength, psi | ----- | ----- | ----- | ----- | ----- | ----- | 11 | 5.5 | 2 | 4 | 7 | 7.8 | ----- | 2 | 5.5 | 4.8 |
| 0.1-percent-offset yield strength, psi | ----- | ----- | ----- | ----- | 127,000 | 131,000 | 139,000 | 138,000 | 110,500 | 132,500 | 136,500 | 133,500 | 105,000 | 116,000 | 123,000 | 119,000 |
| 0.2-percent-offset yield strength, psi | ----- | ----- | ----- | ----- | 85,000 | 78,000 | 79,000 | 81,000 | 76,000 | 72,000 | 82,000 | 75,500 | 71,200 | 58,000 | 67,000 | 65,000 |
| Elongation, percent | ----- | ----- | ----- | ----- | 94,500 | 85,000 | 87,000 | 88,000 | 85,000 | 79,600 | 89,000 | 81,000 | 80,800 | 62,500 | 75,500 | 71,500 |
| | ----- | ----- | ----- | ----- | 6 | 6 | 8.0 | 8.5 | 1 | 3 | 9.0 | 10.7 | 1.5 | 5 | 7.4 | 7.0 |

^aHeat treatments:

NR-74B-F As-forged and aged disc; 16 hr at 1400° F; air-cool.

NR-74B-QA Heat-treated and aged disc; 2300° F, 3 1/4 hr; water-quenched plus 16 hr at 1400° F; air-cool.

NR-76B-F As-forged and aged disc; 16 hr at 1400° F; air-cool.

NR-76B-Q Heat-treated and aged disc; 2300° F, 2 1/2 hr; water-quenched plus 16 hr at 1400° F; air-cool.

^bS-816 disc data taken from reference 10.^cEstimated values.

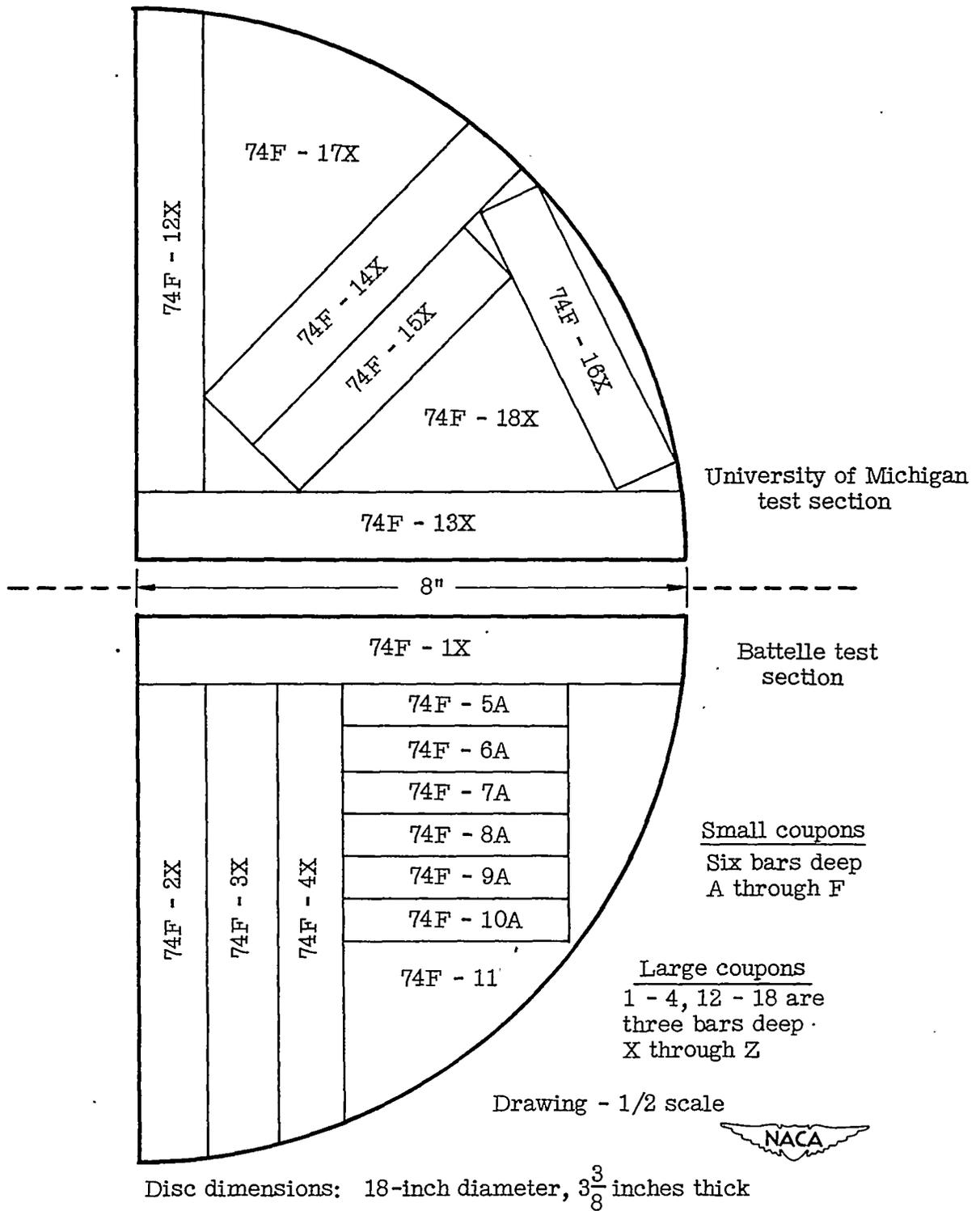
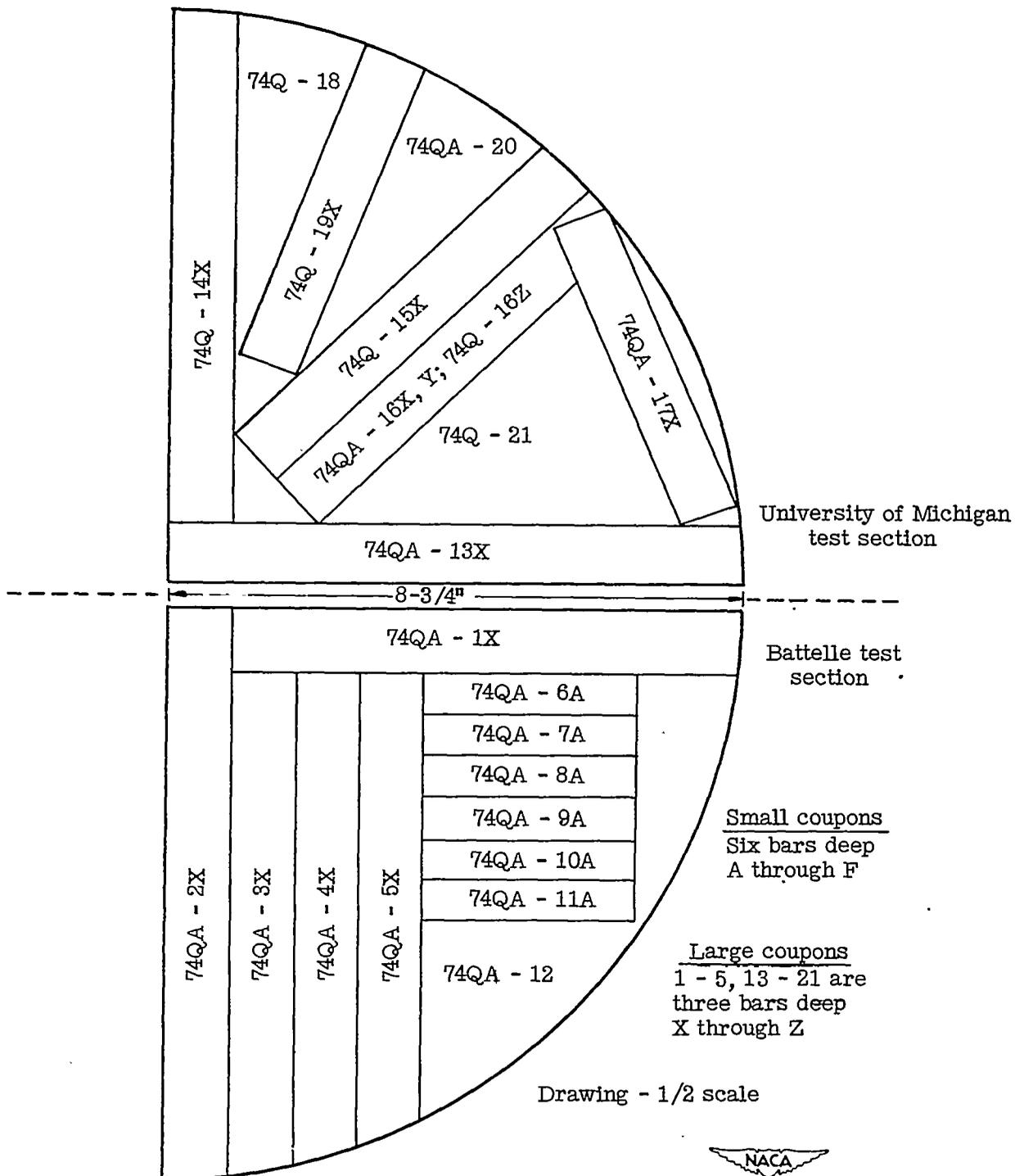


Figure 1.- Location of test coupons in forged and aged S-590 alloy disc NR-74B-F. Coupons: as-forged; 16 hours at 1400° F.



Disc dimensions: 18-inch diameter, 3 inches thick

Figure 2.- Location of test coupons in heat-treated S-590 alloy disc NR-74B-Q. Coupons 74Q: 2300° F, water-quenched. Coupons 74QA: 2300° F, water-quenched; 16 hours at 1400° F.

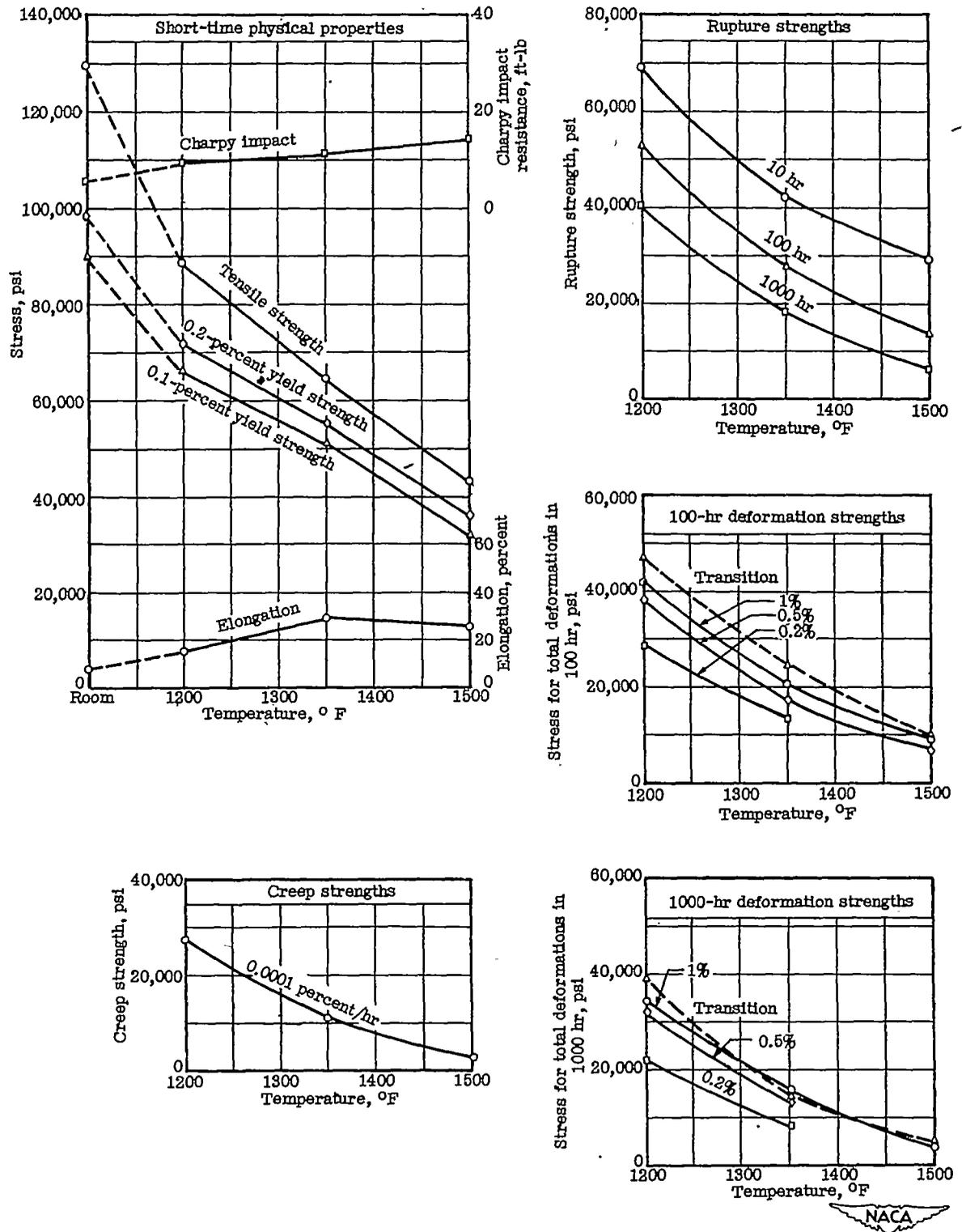


Figure 3.- Summary of properties of S-590 alloy disc NR-74B-F. Disc treatment: as-forged; 16 hours at 1400° F.

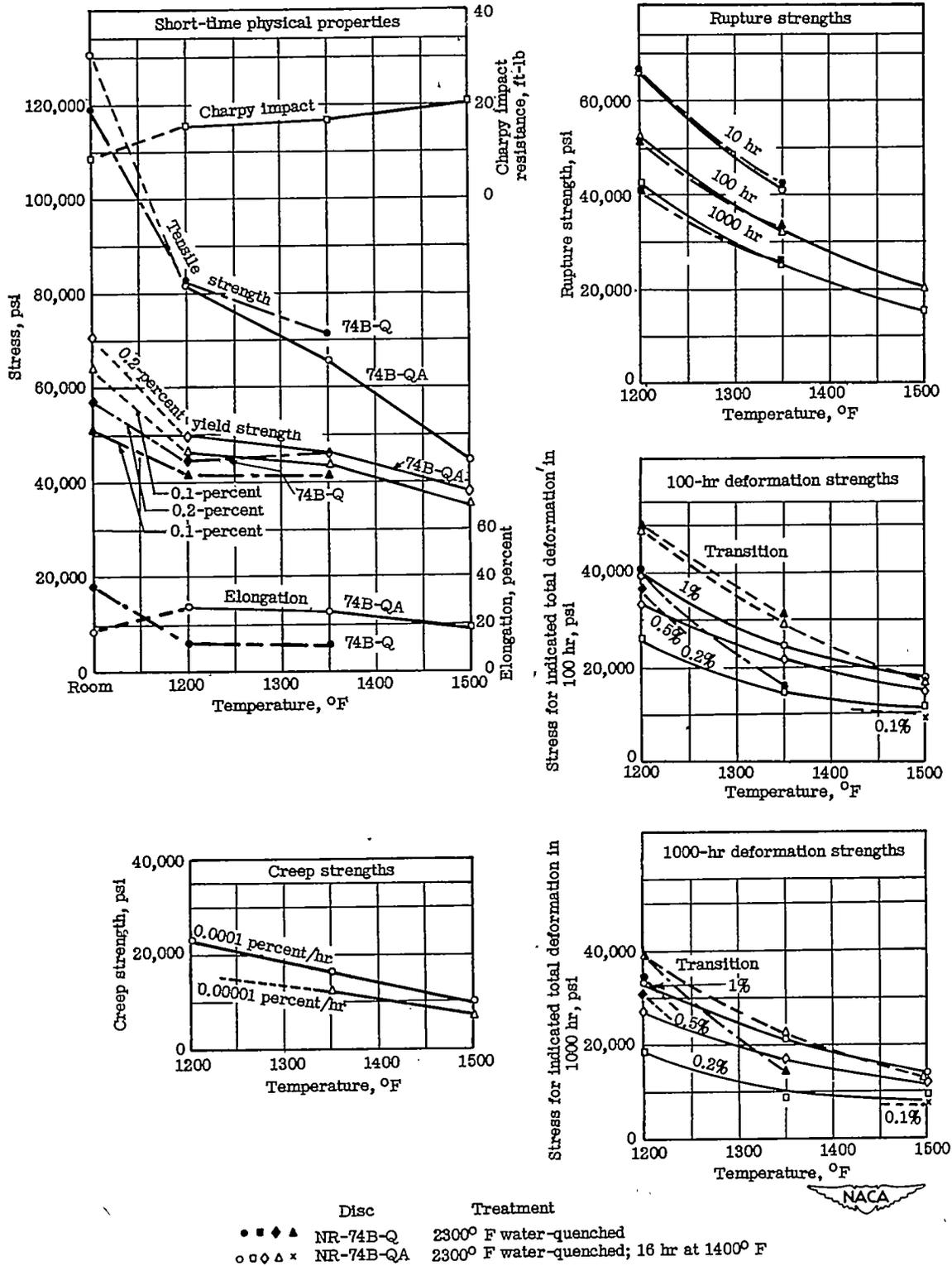


Figure 4.- Summary of properties of S-590 alloy discs NR-74B-Q and NR-74B-QA.

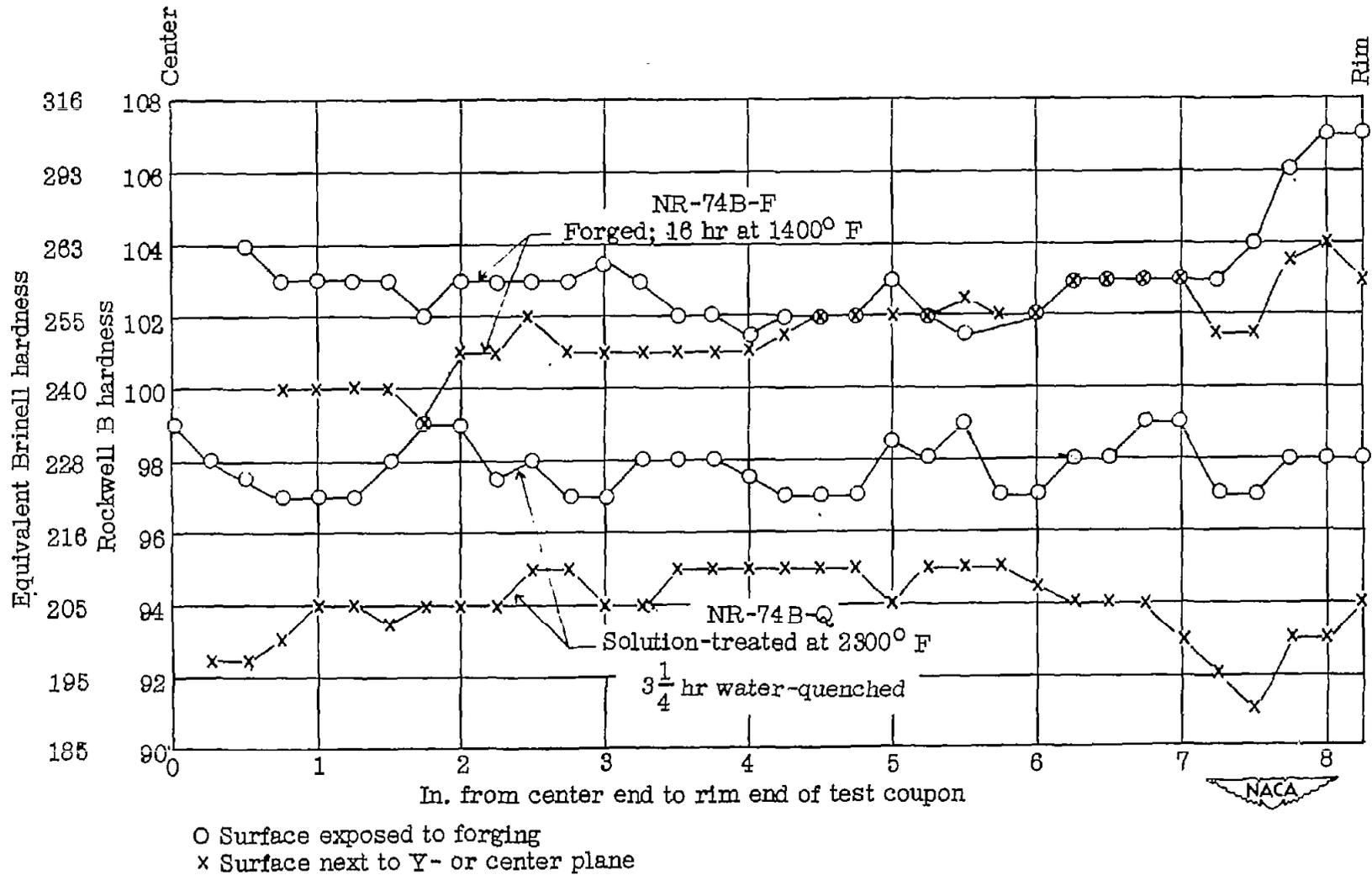


Figure 5.- Variation in hardness from center to rim of S-590 alloy discs NR-74B.

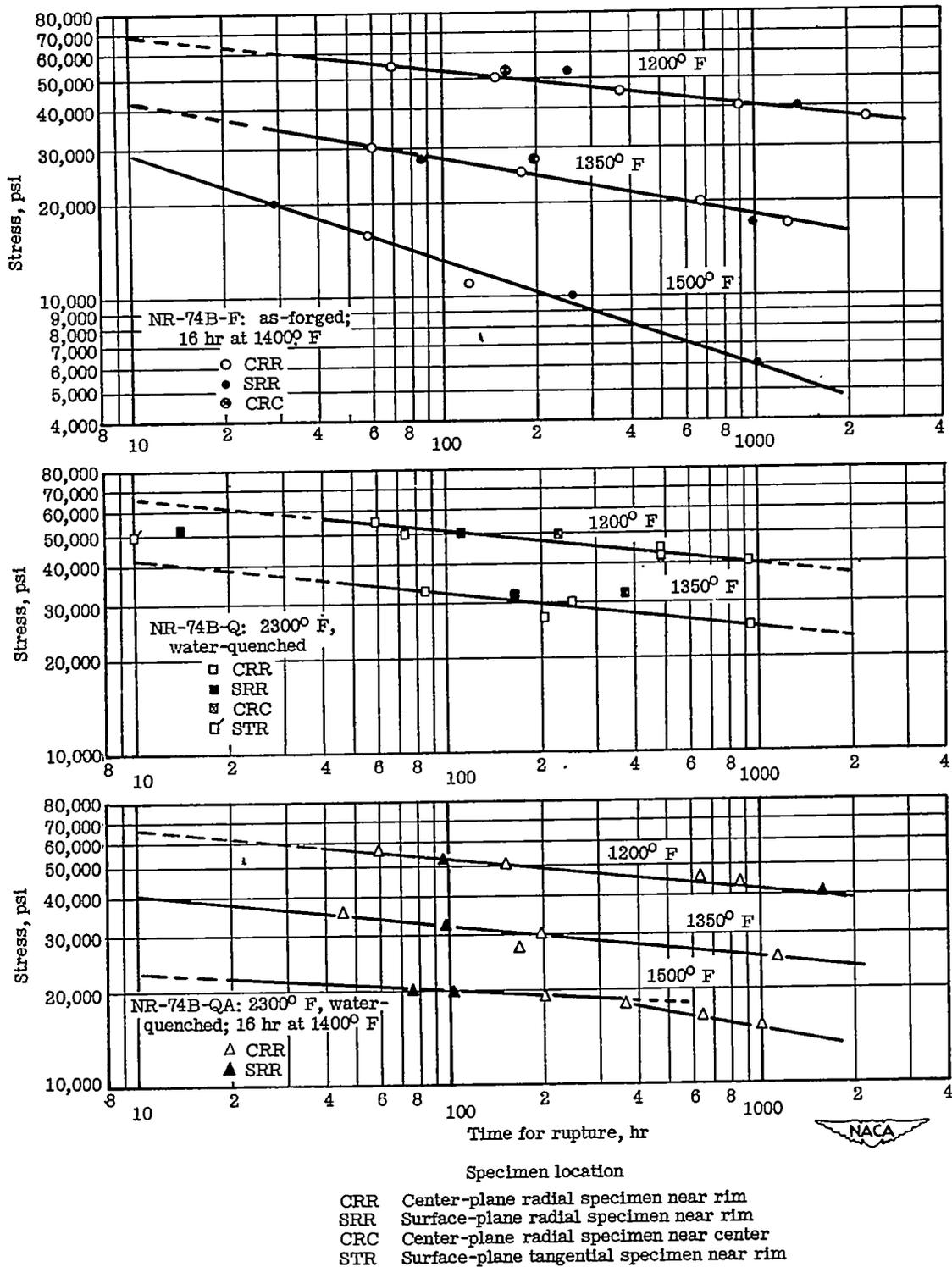


Figure 6:- Curves of stress against rupture time at 1200°, 1350°, and 1500° F for S-590 alloy discs NR-74B.

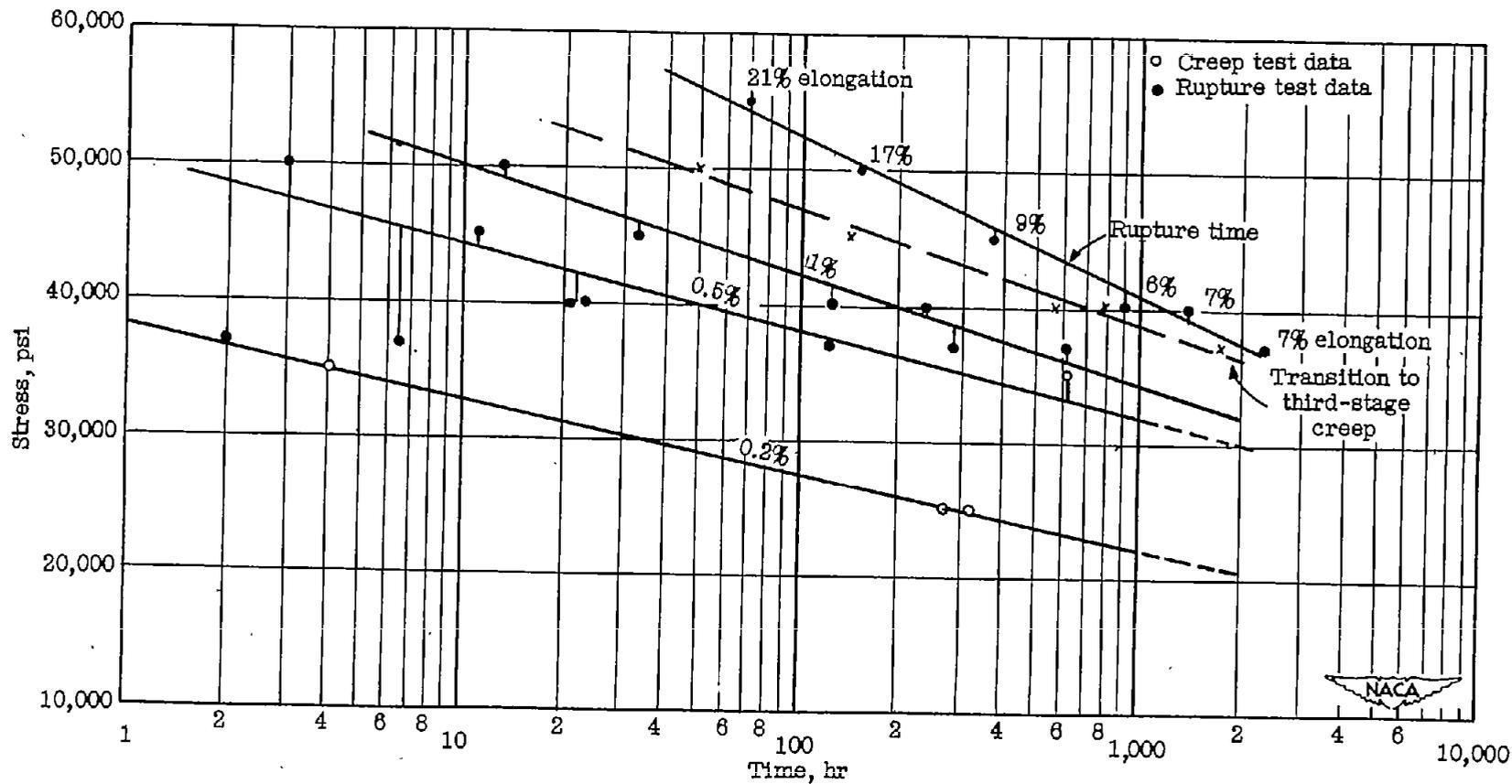


Figure 7.- Curves of stress against time for total deformation at 1200° F for S-590 alloy disc NR-74B-F.
Heat treatment: as-forged; 16 hours at 1400° F.

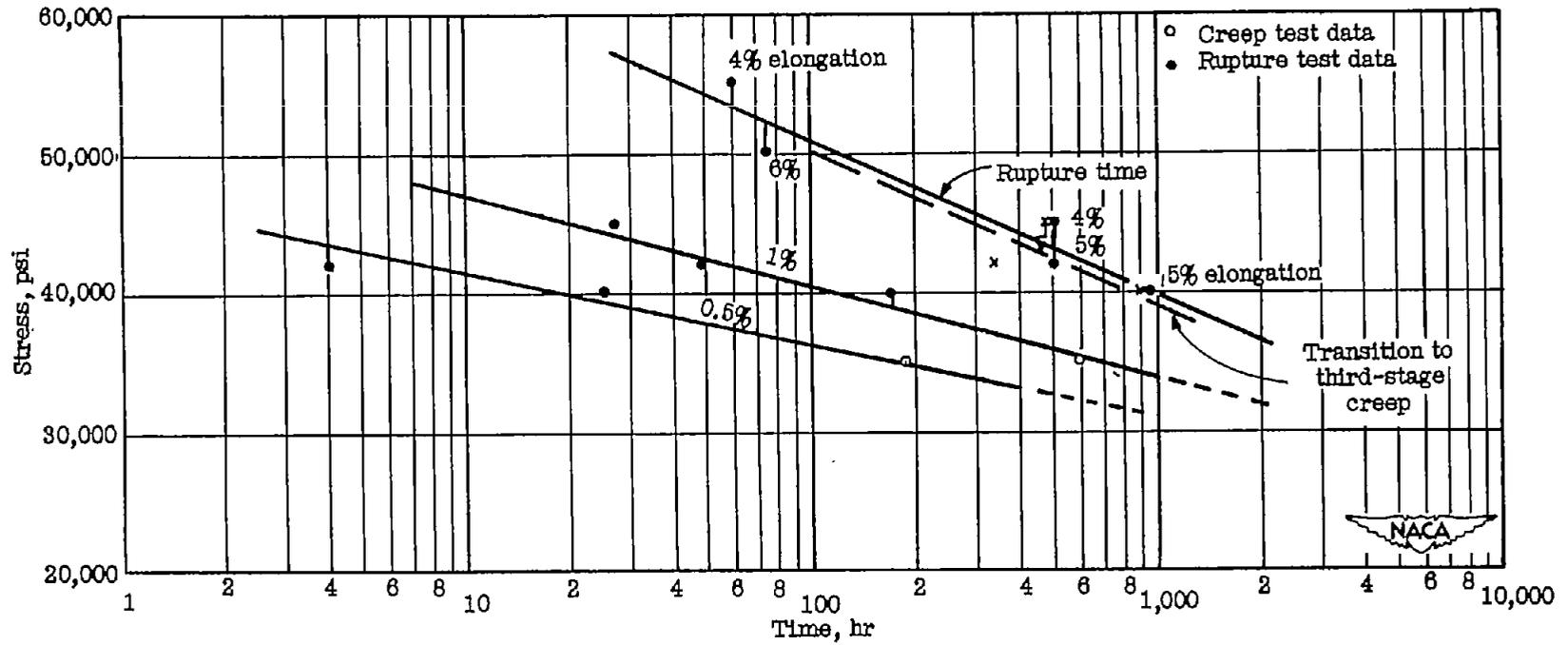


Figure 8.- Curves of stress against time for total deformation at 1200° F for S-590 alloy disc NR-74B-Q.
Heat treatment: 2300° F water-quenched.

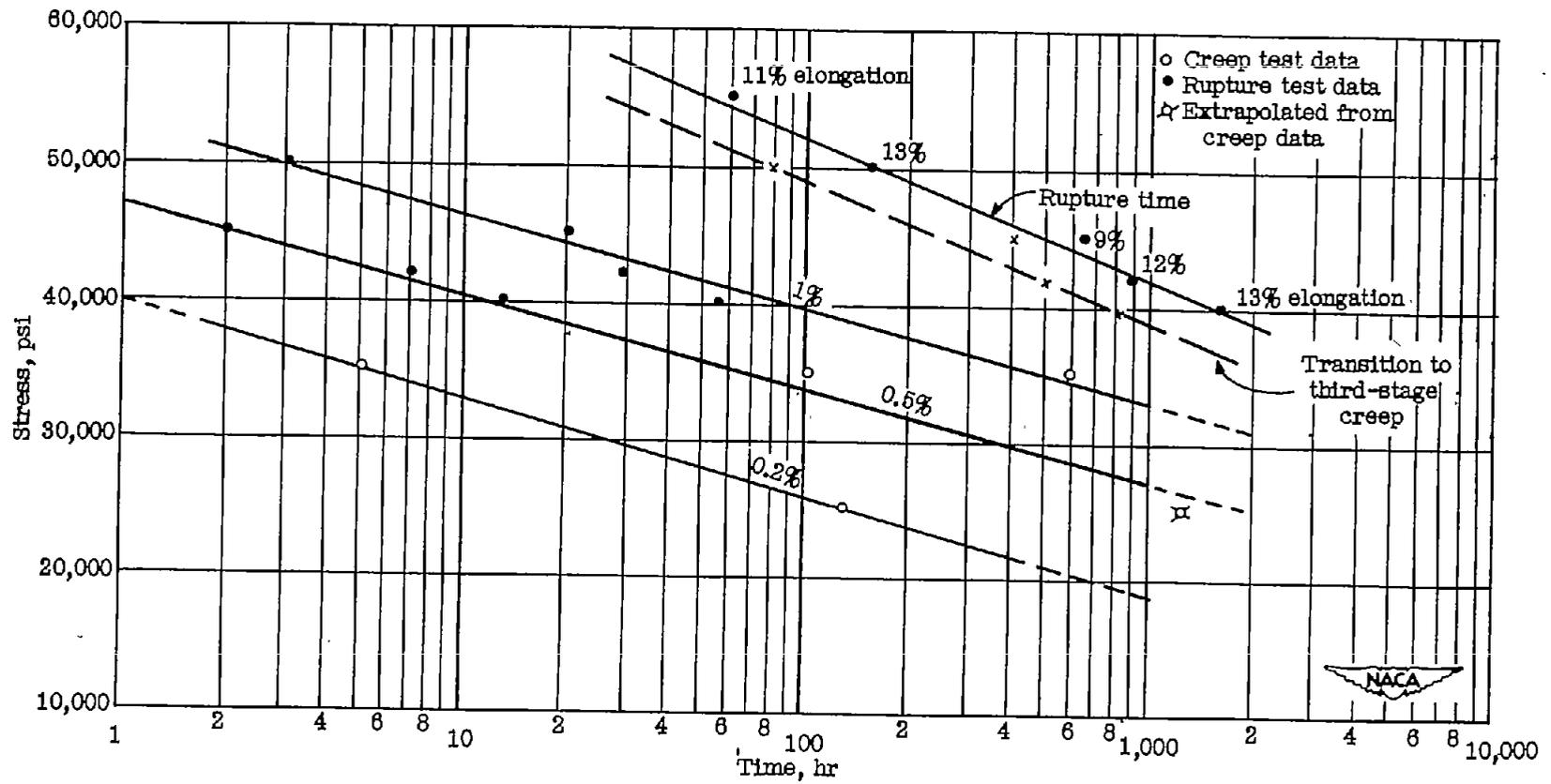


Figure 9.- Curves of stress against time for total deformation at 1200° F for S-590 alloy disc NR-74B-QA.
Heat treatment: 2300° F water-quenched; 16 hours at 1400° F.

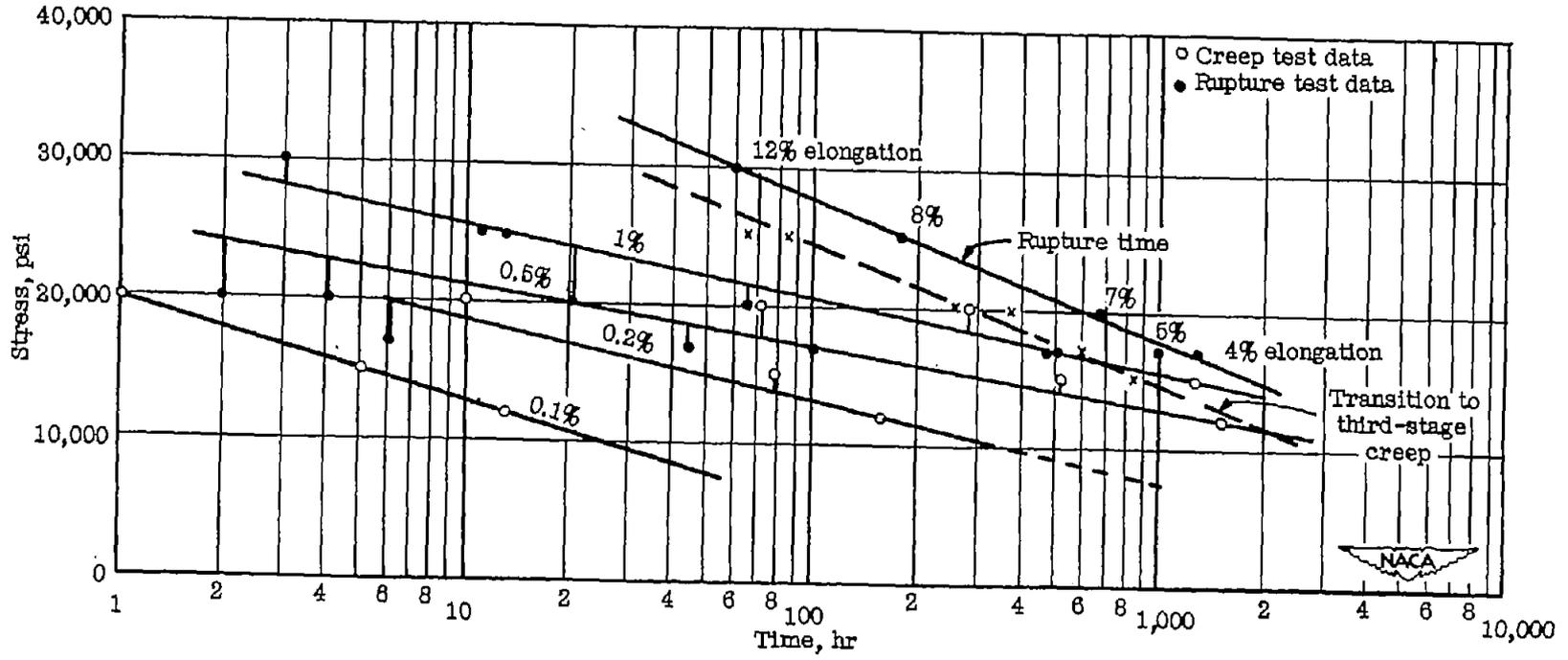


Figure 10.- Curves of stress against time for total deformation at 1350° F for S-590 alloy disc NR-74B-F. Heat treatment: as-forged; 16 hours at 1400° F.

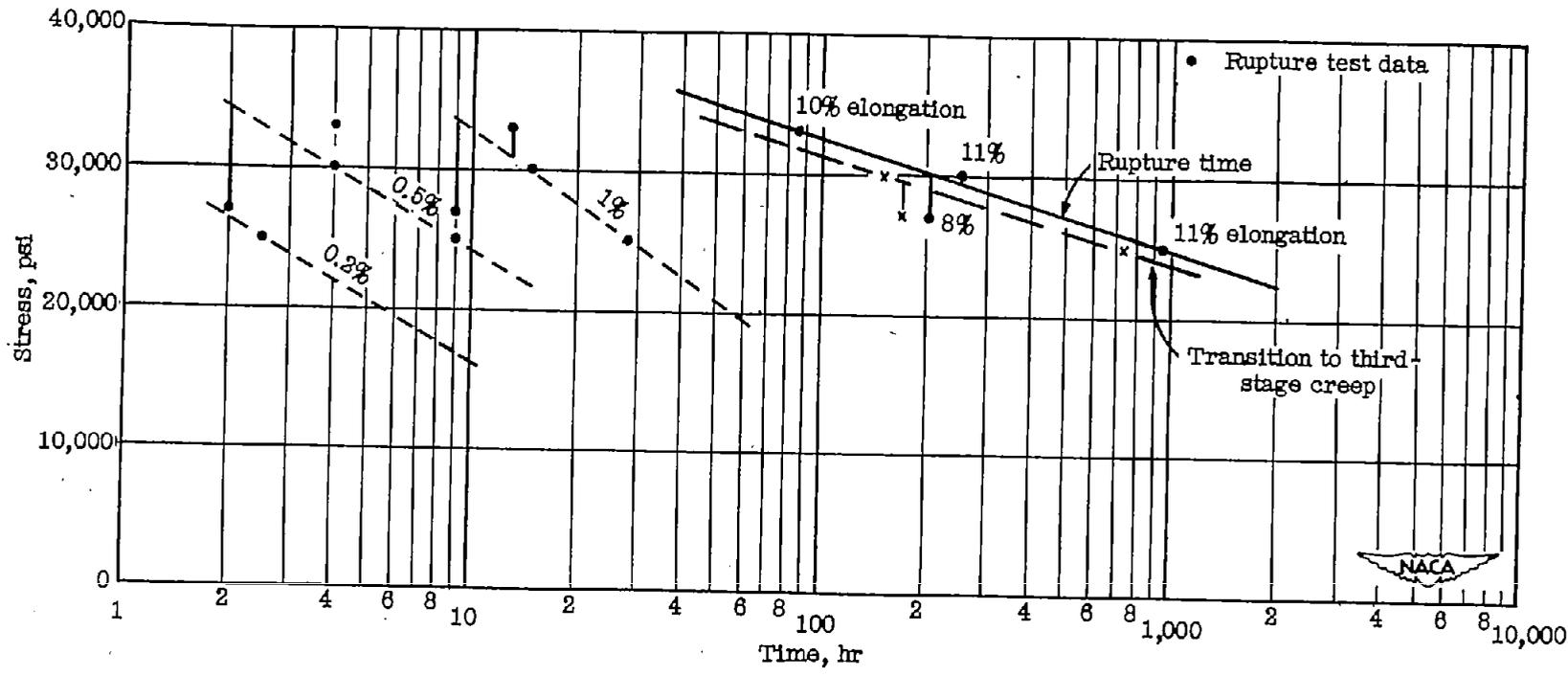


Figure 11.- Curves of stress against time for total deformation at 1350° F for S-590 alloy disc NR-74B-Q. Heat treatment: 2300° F water-quenched.

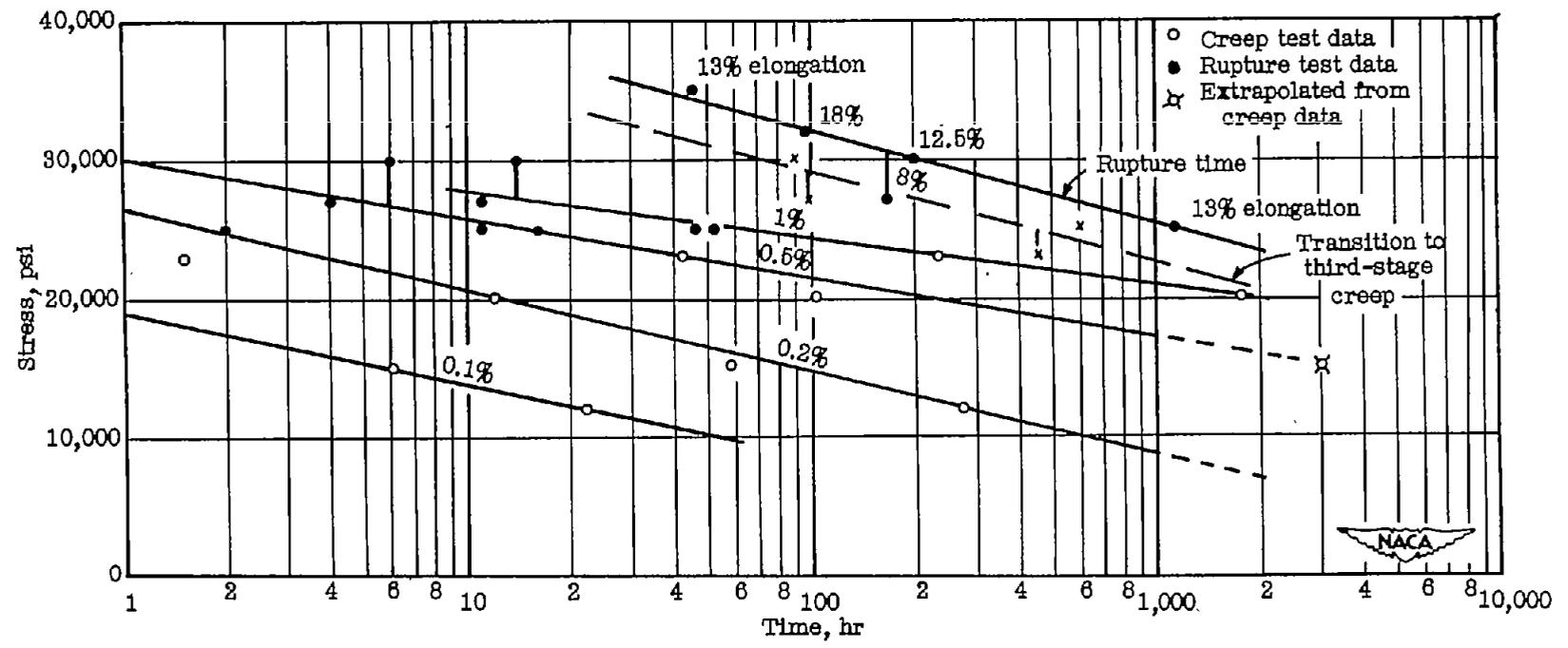


Figure 12.- Curves of stress against time for total deformation at 1350° F for S-590 alloy disc NR-74B-QA. Heat treatment: 2300° F water-quenched; 16 hours at 1400° F.

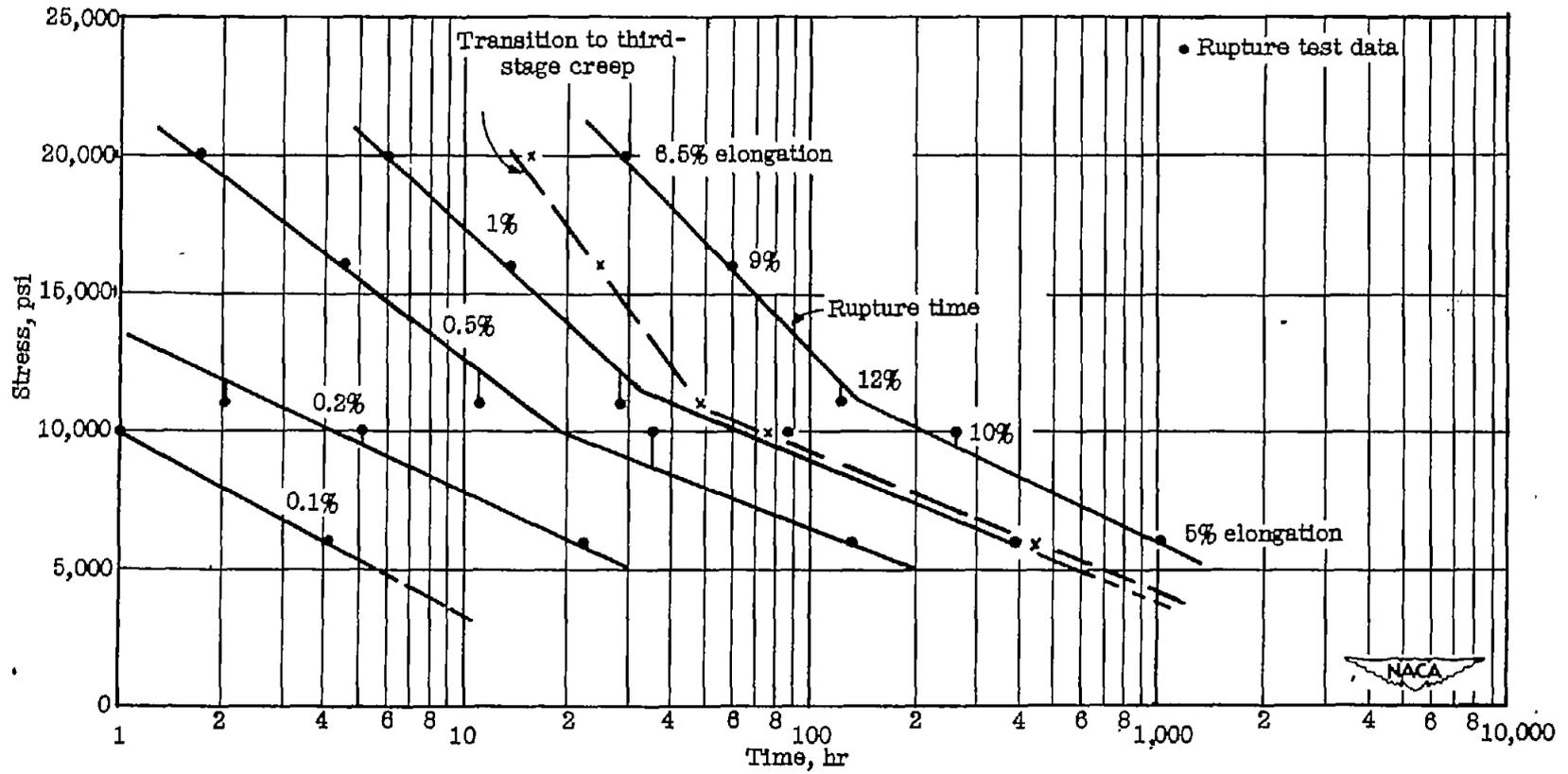


Figure 19.- Curves of stress against time for total deformation at 1500° F for S-590 alloy disc NR-74B-F.
Heat treatment: as-forged; 16 hours at 1400° F.

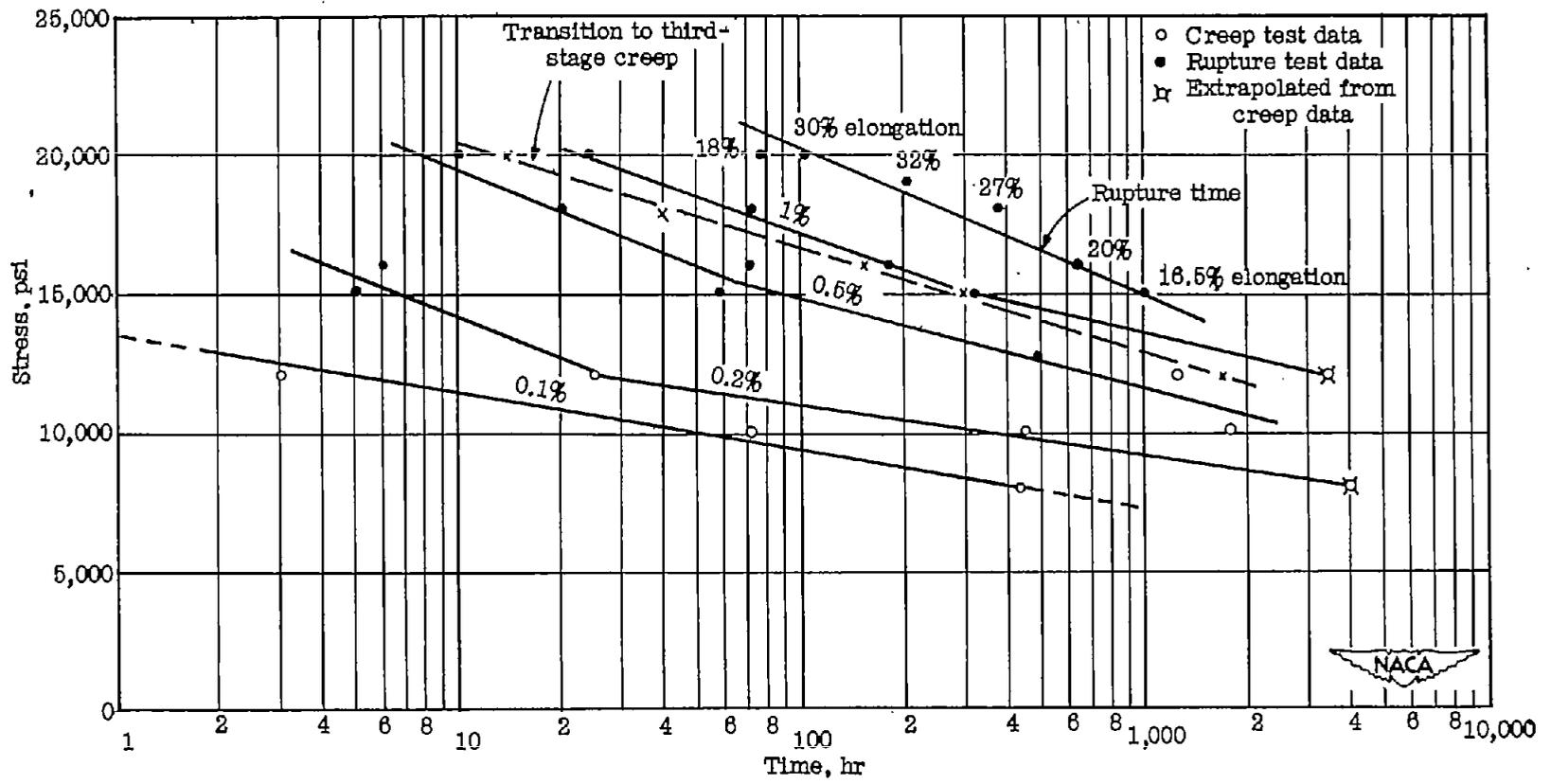


Figure 14.- Curves of stress against time for total deformation at 1500° F for S-590 alloy disc NR-74B-QA. Heat treatment: 2300° F water-quenched; 16 hours at 1400° F.

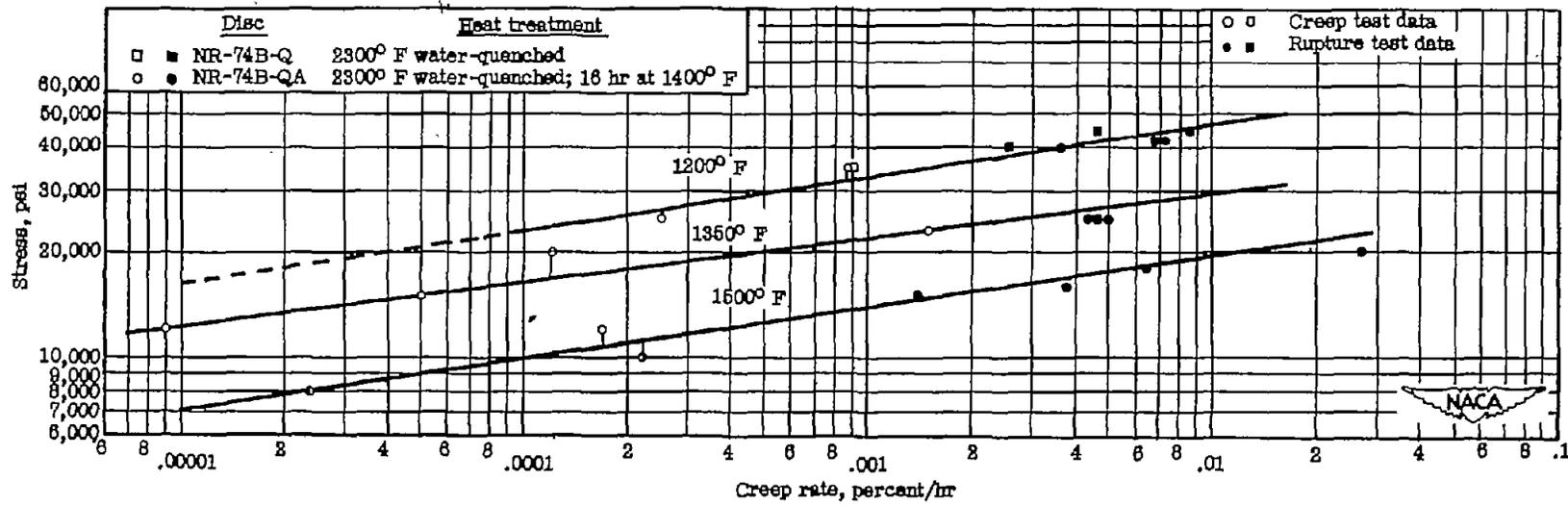
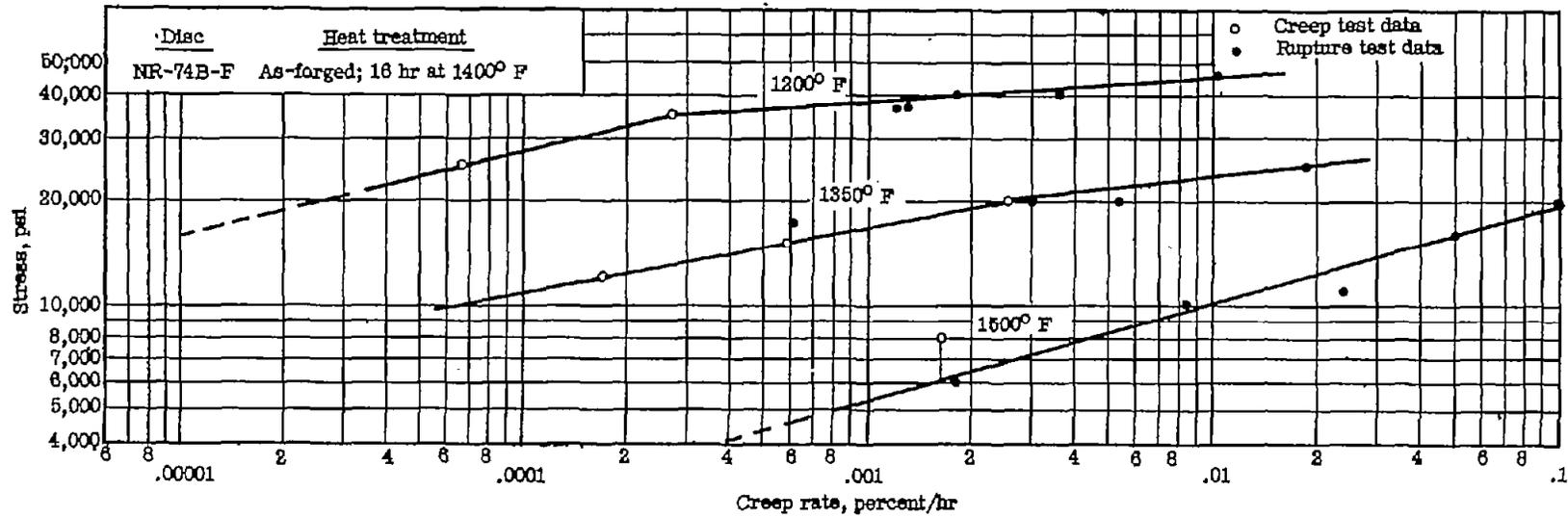
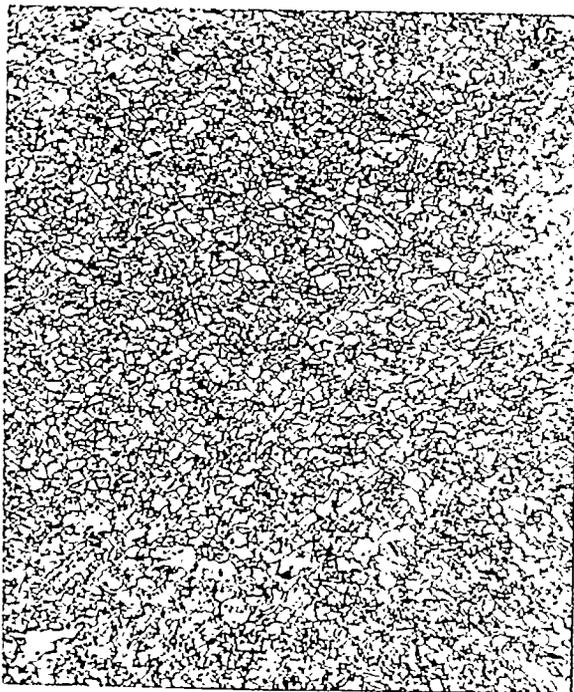


Figure 15.- Curves of stress against creep rate at 1200°, 1350°, and 1500° F for S-590 alloy discs NR-74B.



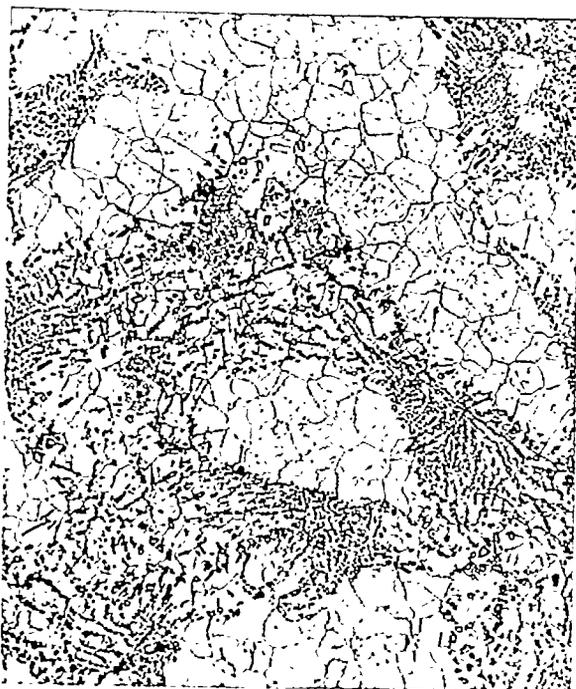


100X



1000X

(a) Radial section near rim of disc in center plane.



100X



1000X

(b) Radial section near center of disc in center plane.

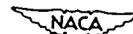


Figure 16.- Original microstructure of S-590 alloy disc NR-74B-F.
Electrolytic chromic acid etch. Disc treatment: as-forged; 16 hours
at 1400° F.



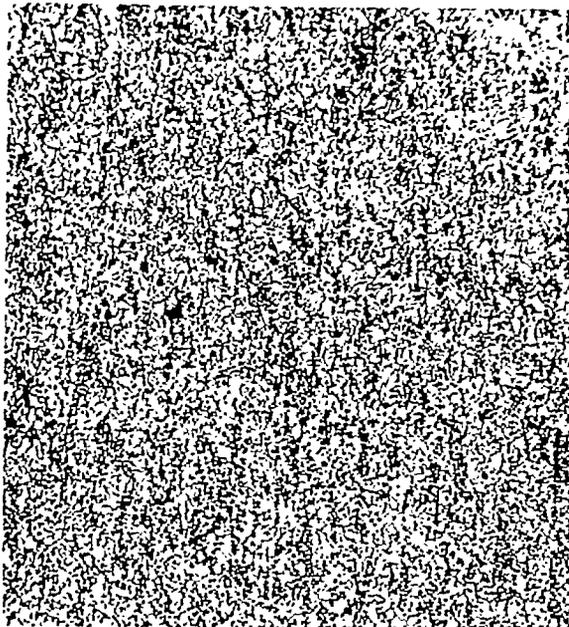


100X



1000X

(a) Specimen 13X; 1002 hours at 1200° F under 35,000 psi.



100X

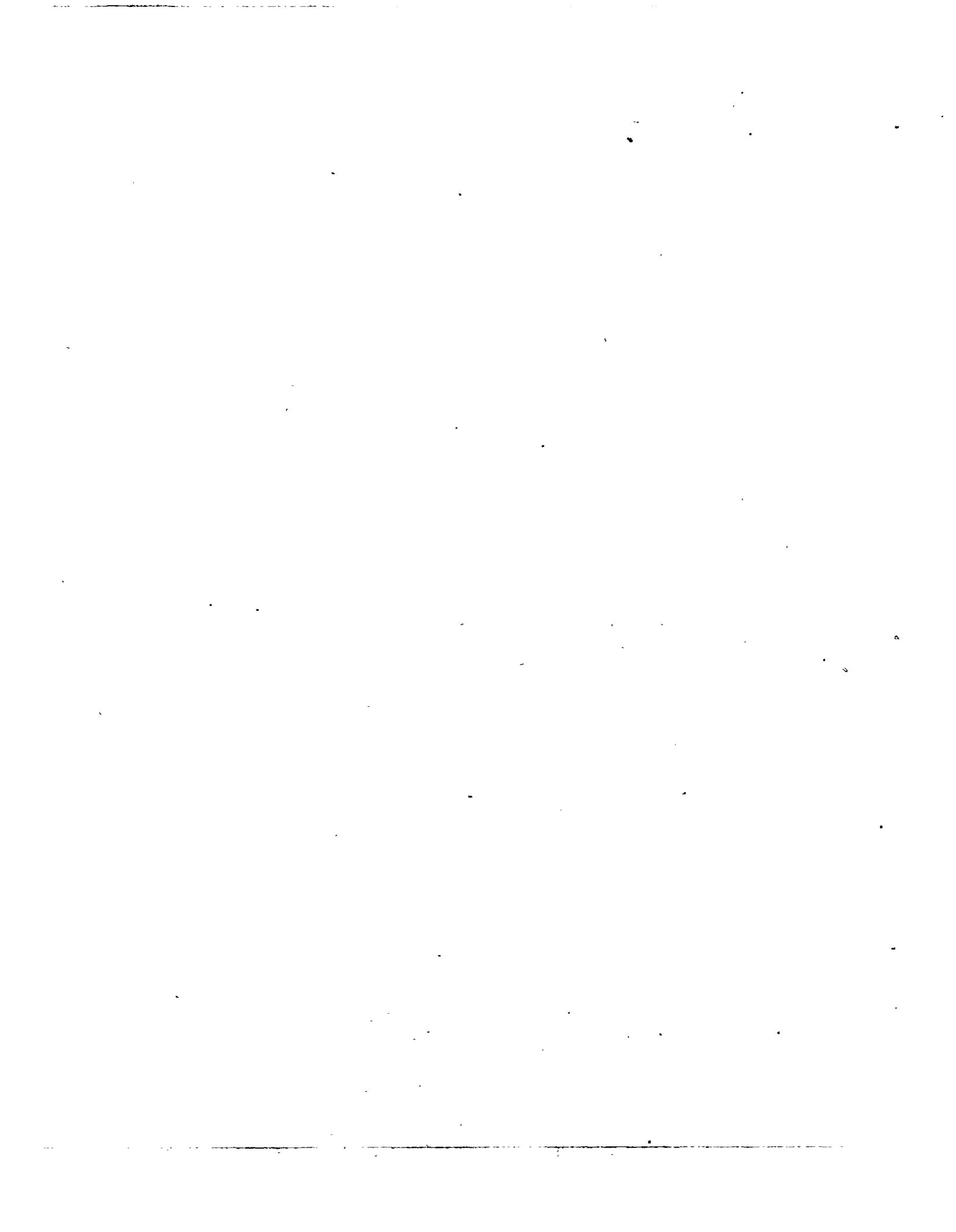


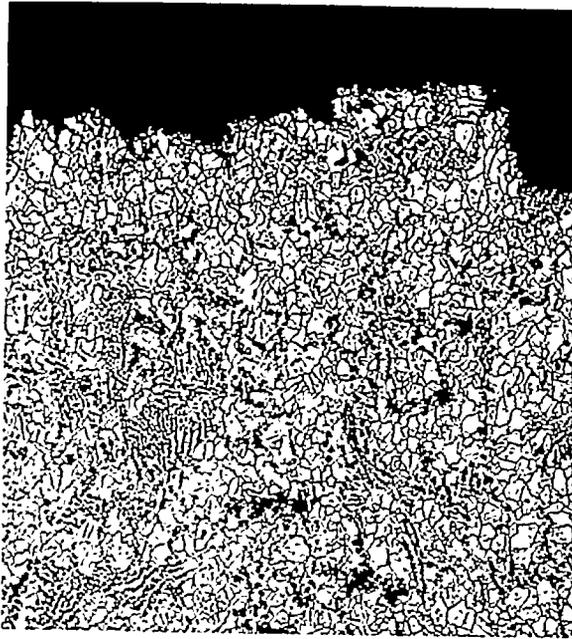
1000X

(b) Specimen 2X; 1872 hours at 1350° F under 15,000 psi.



Figure 17.- Microstructure of specimens from S-590 alloy disc NR-74B-F after creep tests. Electrolytic chromic acid etch. Disc treatment: as-forged; 16 hours at 1400° F.





Fracture - 100X



Interior - 1000X

(a) Specimen 12Y; 2310 hours for rupture at 1200° F under 37,000 psi.



Fracture - 100X



Interior - 1000X

(b) Specimen 12Y; 1291 hours for rupture at 1350° F under 17,000 psi.

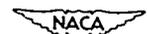
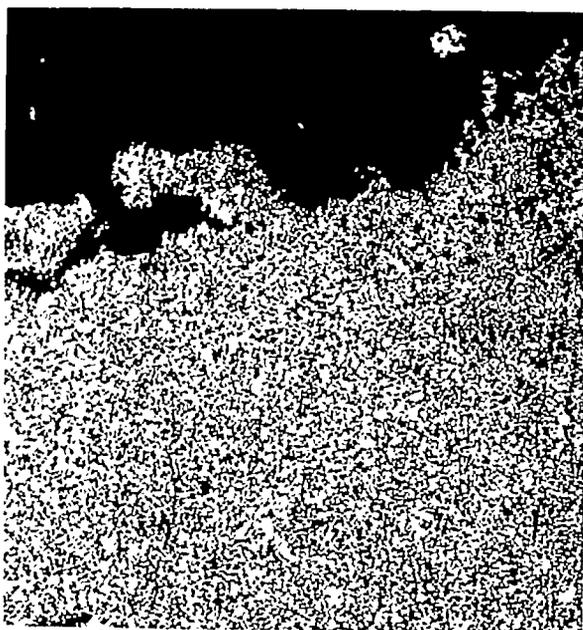
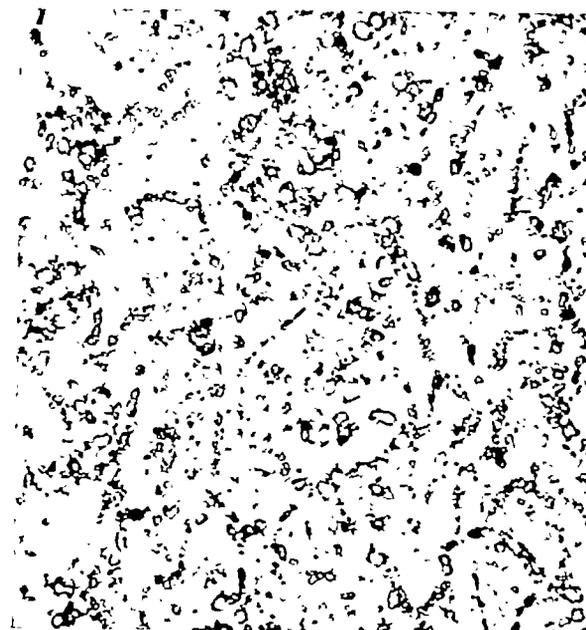


Figure 18.- Microstructure of specimens from S-590 alloy disc NR-74B-F after stress-rupture tests. Electrolytic chromic acid etch. Disc treatment: as-forged; 16 hours at 1400° F.





Fracture - 100X



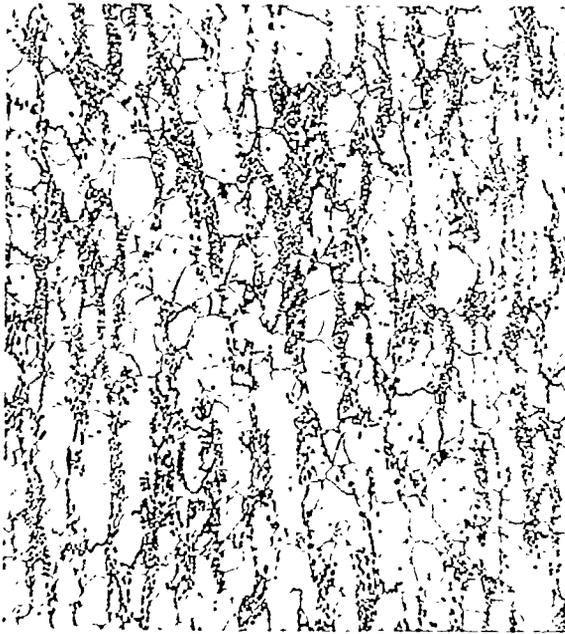
Interior - 1000X

(c) Specimen 9F; 1018 hours for rupture at 1500° F under 6000 psi.

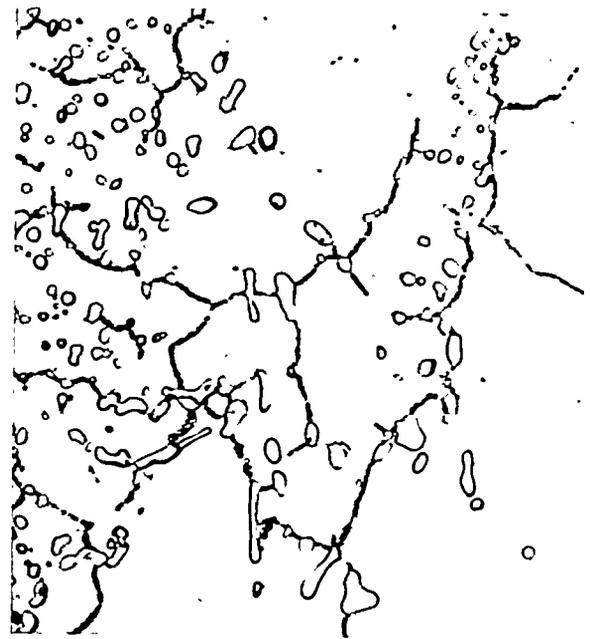


Figure 18.- Concluded.



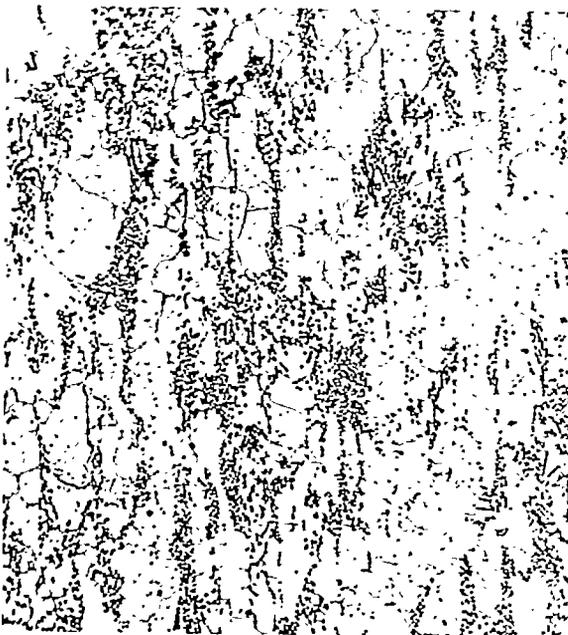


100X



1000X

(a) Disc NR-74B-Q; radial section near rim of disc in center plane. Electrolytic chromic acid etch. Disc treatment: 2300° F water-quenched.



100X



1000X

(b) Disc NR-74B-QA; radial section near rim of disc in center plane. Electrolytic sodium cyanide etch. Disc treatment: 2300° F water-quenched; 16 hours at 1400° F.

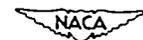
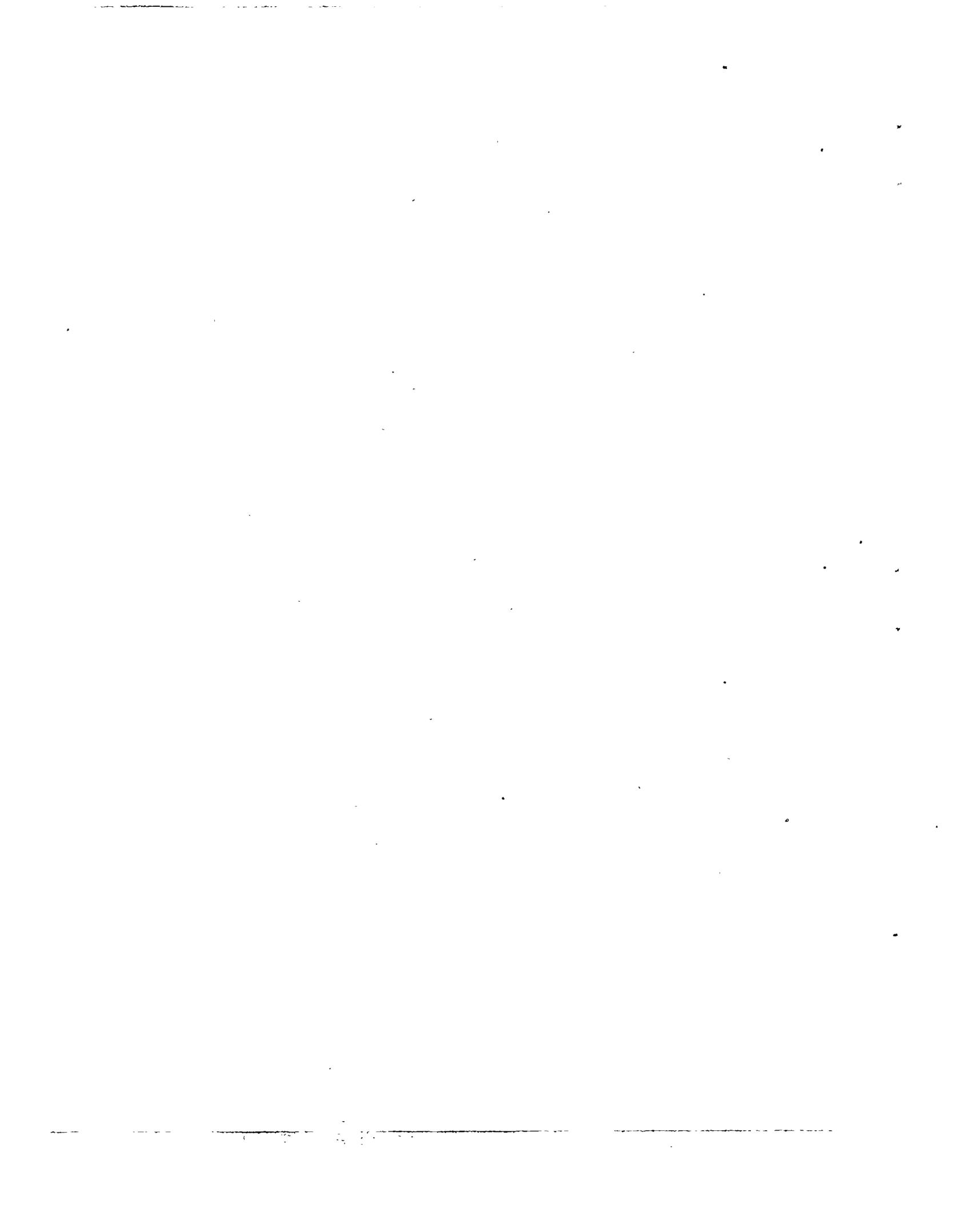
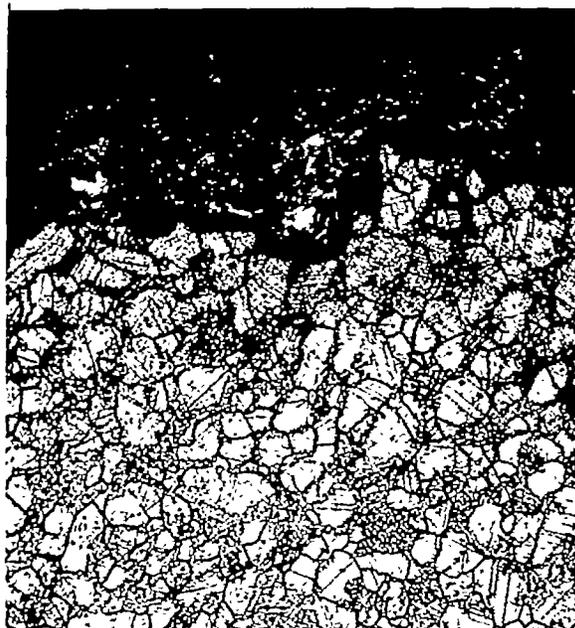


Figure 19.- Original microstructure of S-590 alloy discs NR-74B-Q and NR-74B-QA.



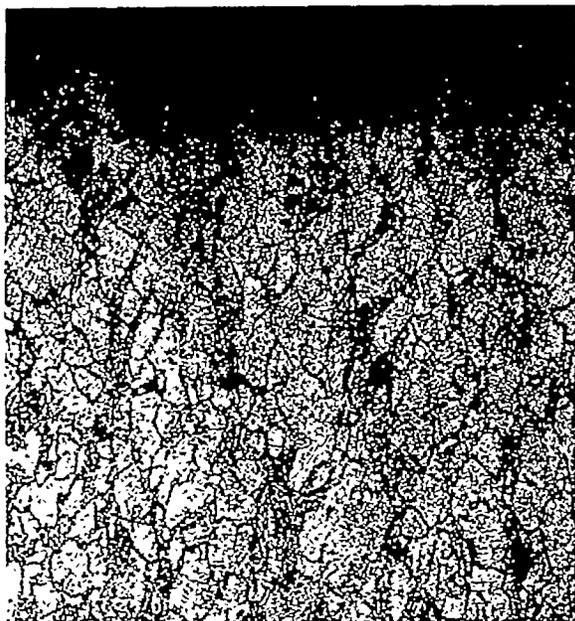


Fracture - 100X



Interior - 1000X

(a) Specimen 18Y; 937 hours for rupture at 1200° F under 40,000 psi.



Fracture - 100X



Interior - 1000X

(b) Specimen 14Y; 951 hours for rupture at 1350° F under 25,000 psi.

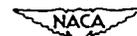
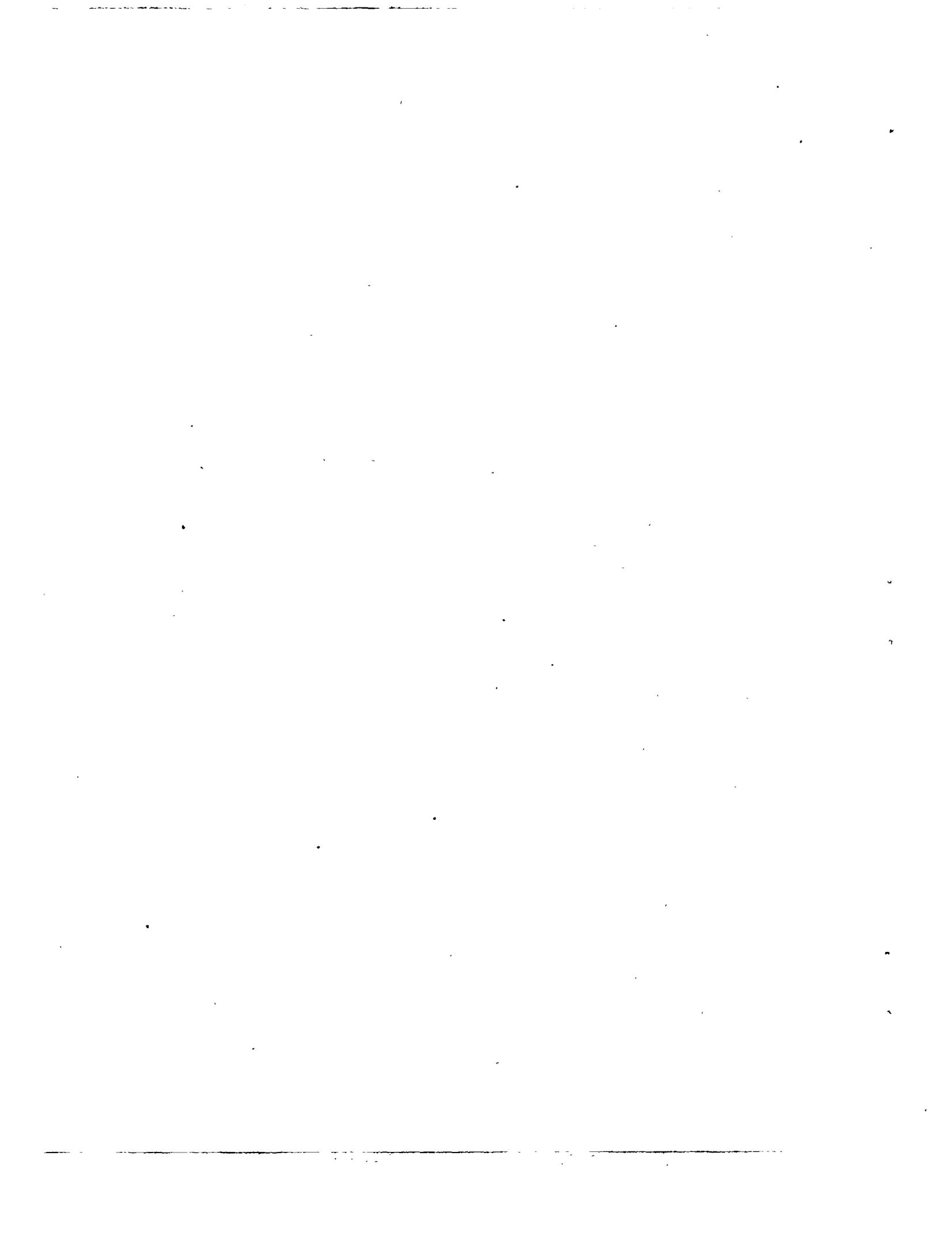
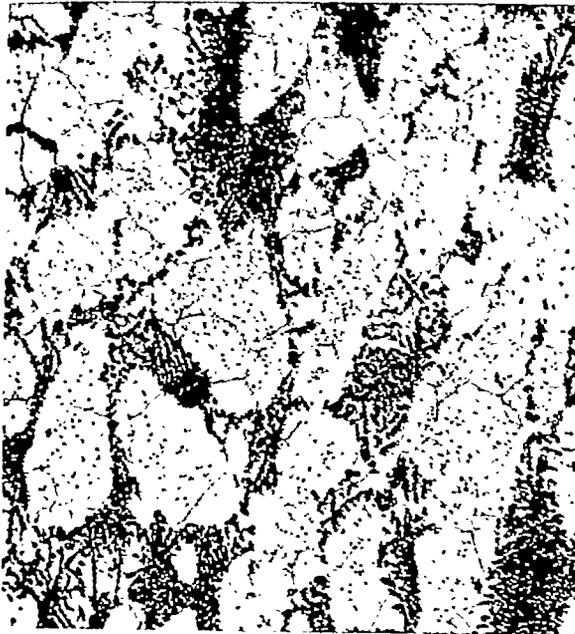
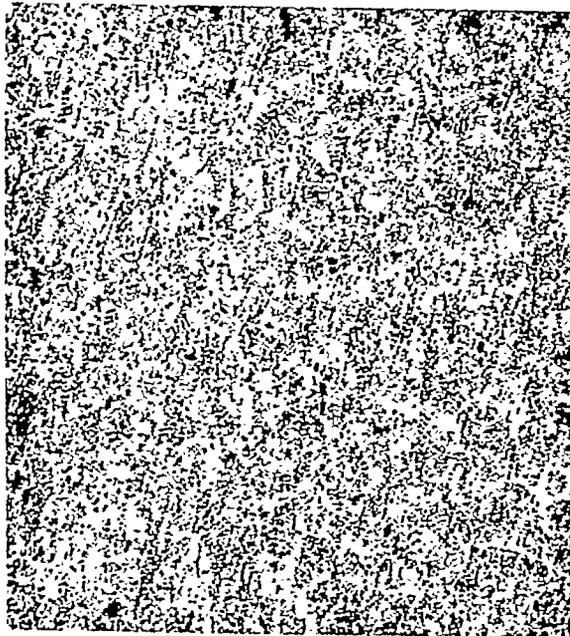


Figure 20.- Microstructure of specimens from S-590 alloy disc NR-74B-Q after stress-rupture tests. Electrolytic sodium cyanide etch. Disc treatment: 2300° F water-quenched.





100X
 (a) Specimen 13Z; 1002 hours at 1200° F under 35,000 psi. 1000X

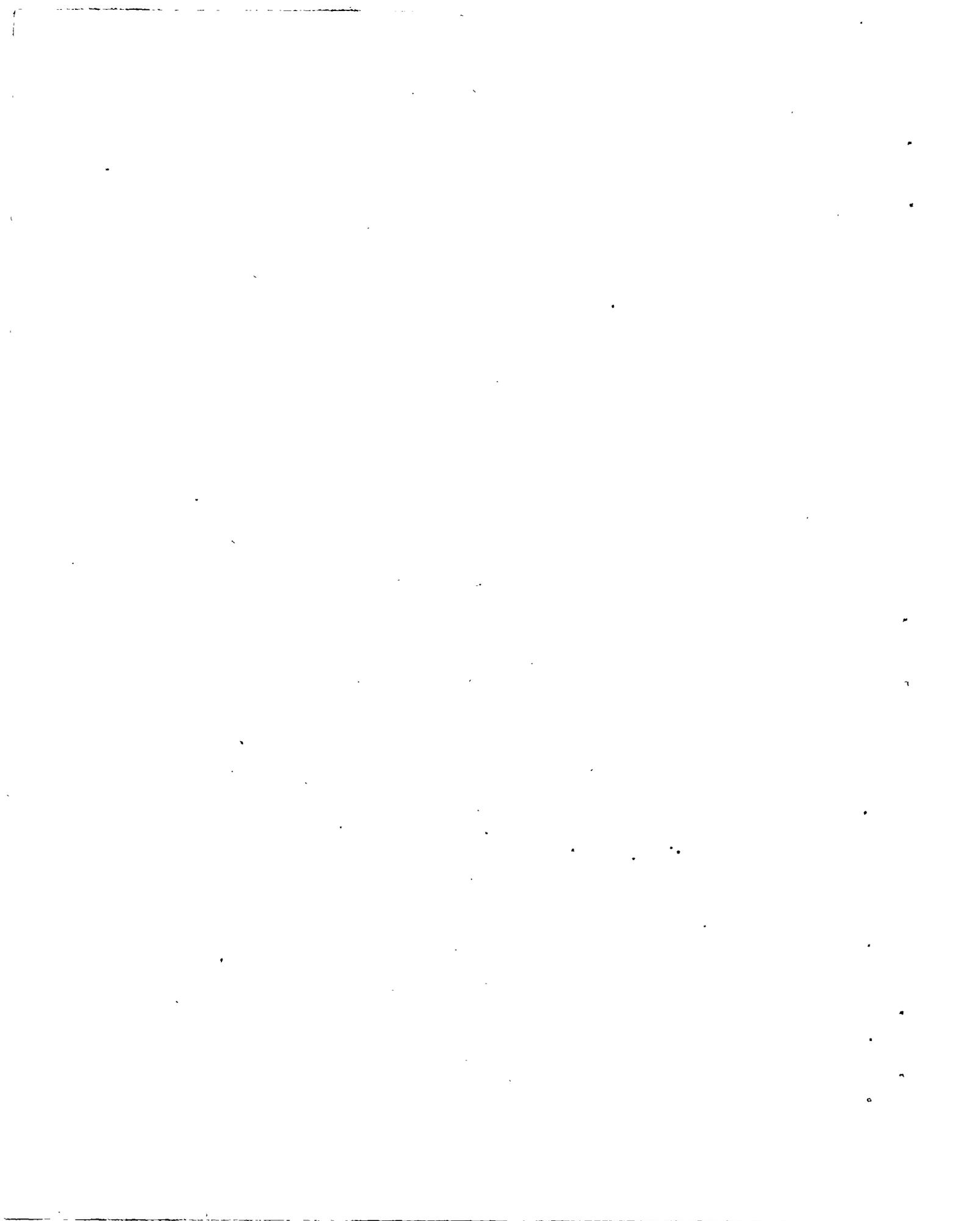


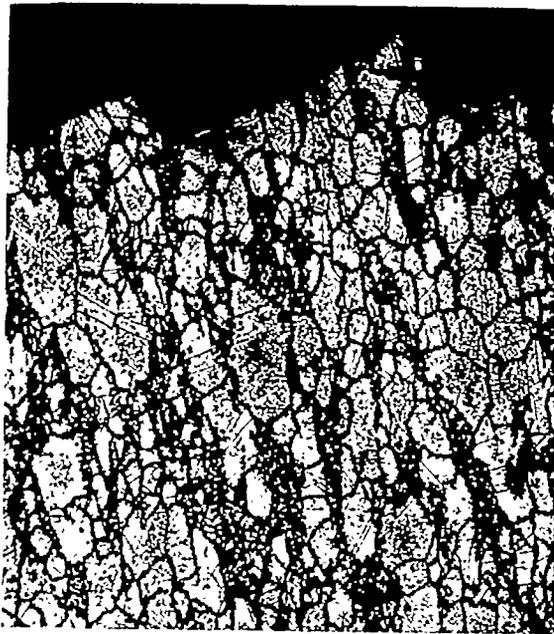
100X
 (b) Specimen 1X; 2282 hours at 1350° F under 15,000 psi. 1000X

Figure 21.- Microstructure of specimens from S-590 alloy disc NR-74B-QA after creep tests. Electrolytic chromic acid etch. Disc treatment: 2300° F water-quenched; 16 hours at 1400° F.

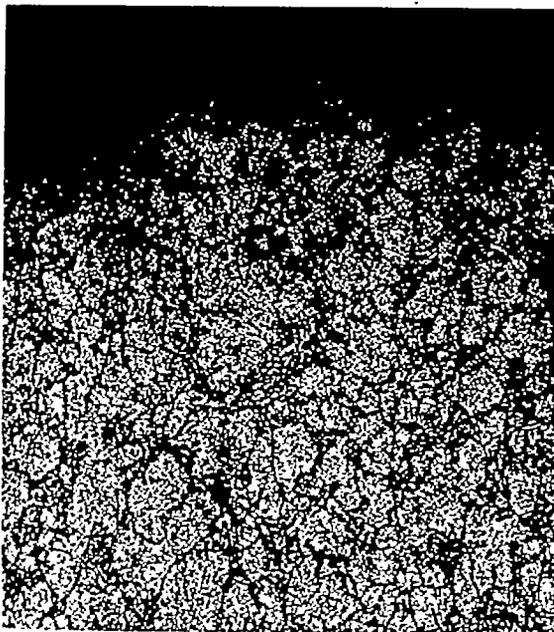








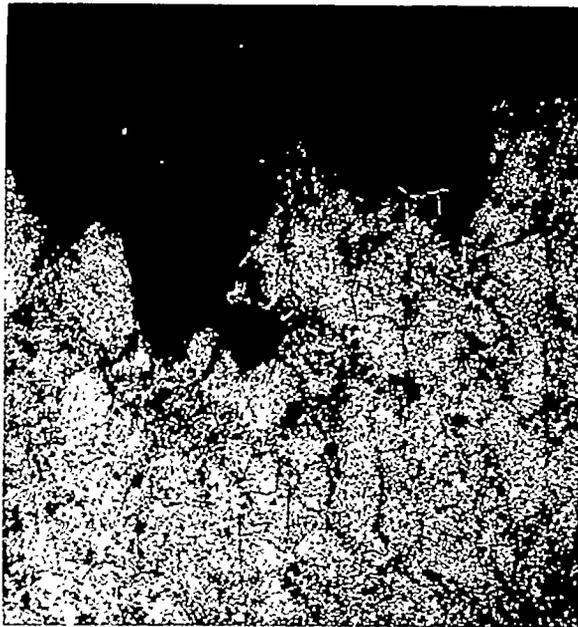
Fracture - 100X
 Interior - 1000X
 (a) Specimen 13X; 1596 hours for rupture at 1200° F under 40,000 psi.



Fracture - 100X
 Interior - 1000X
 (b) Specimen 13Y; 1121 hours for rupture at 1350° F under 25,000 psi.



Figure 22.- Microstructure of specimens from S-590 alloy disc NR-74B-QA after stress-rupture tests. Electrolytic chromic acid etch. Disc treatment: 2300° F water-quenched; 16 hours at 1400° F.



Fracture - 100X



Interior - 1000X

(c) Specimen 11D; 1000 hours for rupture at 1500° F under 15,000 psi.



Figure 22.- Concluded.

TABLE VII

DATA ON STRESS AND TIME FOR TOTAL DEFORMATION AT 1500° F FOR S-590 ALLOY DISCS NR-74B

[NDRC and Navy data]

| Disc (a) | Specimen number | Stress (psi) | Initial deformation (percent) | Time (hr) for total deformations of- | | | | | | Transition to third-stage creep | |
|-------------|--------------------|-----------------|-------------------------------------|--------------------------------------|-------------------|-------------|-------------------|-----------|-----------|------------------------------------|--------------------------|
| | | | | 0.1 percent | 0.2 percent | 0.5 percent | 1 percent | 2 percent | 5 percent | Time (hr) | Deformation (percent) |
| NR-74B-F | 9A | 20,000 | ----- | --- | ----- | 1.7 | 6 | 16 | 28 | 15.5 | 1.95 |
| | 9D | 16,000 | ----- | --- | ----- | 4.5 | 13.5 | 31 | 54 | 24 | 1.56 |
| | 9E | 11,000 | ----- | --- | 2 | 11 | 28 | 64 | 101 | 48 | 1.42 |
| | 2Z | 10,000 | 0.069 | 1 | 5 | 35 | 86 | 145 | 250 | 74 | 0.92 |
| | 9F | 6,000 | ----- | 4 | 22 | 132 | 392 | 700 | --- | 464 | 1.17 |
| NR-74B-QA | 11F | 20,000 | ----- | --- | ----- | 10 | 24 | 42 | 69 | 14 | .60 |
| | 11C | 18,000 | ----- | --- | ----- | 20 | 72 | 119 | 220 | 40 | .62 |
| | 11B | 16,000 | ----- | --- | 6 | 70 | 180 | 306 | 474 | 155 | .85 |
| | 11D | 15,000 | ----- | --- | 5 | 58 | 325 | 545 | 787 | 300 | .94 |
| | 4X | 12,000 | .077 | 3 | 25 | 1270 | ^b 3400 | --- | --- | 1700 | .58 |
| | 1Z | 10,000 | .068 | 72 | 456 | 1800 | ----- | --- | --- | ----- | ----- |
| | 2X | 8,000 | .036 | 430 | ^b 4000 | ----- | ----- | --- | --- | ----- | ----- |

^aHeat treatments:

NR-74B-F As-forged; 16 hr at 1400° F.

NR-74B-QA 2300° F water-quenched; 16 hr at 1400° F.

^bEstimated.